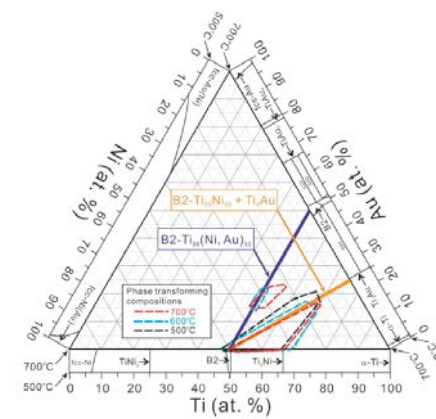
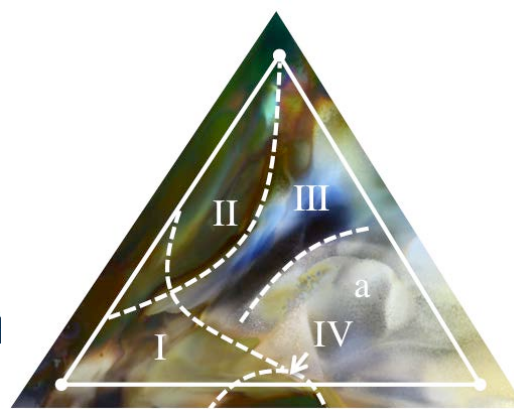
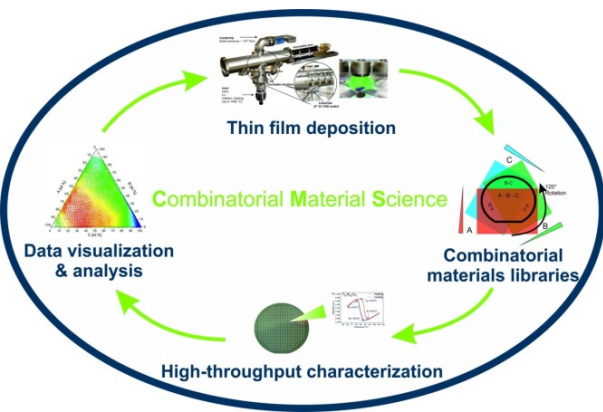




Fundamental Aspects of Materials Science and Engineering

Sommersemester 2020

Phase diagrams, intermetallic phases and combinatorial materials research

Prof. Dr.-Ing. Alfred Ludwig
Materials Discovery and Interfaces

Intermetallics:

Some past and present applications

Since approx.	Material or process	Phase	Application	Reference
2500 B. C.	cementation	Cu_3As	coating of bronze tools, etc. (Egypt, Anatolia, Britain)	Westbrook (1977)
100 B. C.	yellow brass	CuZn	coins, ornamental parts (Rome)	Gmelin-Institut (1955)
0	high tin bronze	$\text{Cu}_{31}\text{Sn}_8$	mirror (China)	Westbrook (1977)
600	amalgam	$\text{Ag}_2\text{Hg}_3 + \text{Sn}_6\text{Hg}$	dental restorative (China)	Westbrook (1977) Waterstrat (1990)
1500	amalgam	Cu_4Hg_3	dental restorative (Germany)	Pauller (1976), Westbrook (1977) Waterstrat (1990)
1505	amalgam	Sn_8Hg	mirror surface (Venice)	Westbrook (1977)
1540	type metal	SbSn	printing	Westbrook (1977)
1910	Acutal	$(\text{Cu}, \text{Mn})_3\text{Al}$	fruit knife (Germany)	Heusler (1989)
1921	Permalloy	Ni_3Fe	high permeability magnetic alloy	Bozorth (1951)
1926	Permendur	$\text{FeCo}(-2\text{V})$	soft magnetic alloy	Bozorth (1951), Chen (1961)
1931	Alnico	$\text{NiAl}-\text{Fe}-\text{Co}$	permanent magnet material	De Vos (1969)
1935	Sendust	$\text{Fe}_3(\text{Si}, \text{Al})$	magnetic head material	Yamamoto (1980)
1938	$\text{Cu}-\text{Zn}-\text{Al}$ $\text{Cu}-\text{Al}-\text{Ni}$	$\text{CuZn}-\text{Al}$ $(\text{Cu}, \text{Ni})_3\text{Al}$	shape memory alloys	Hodgson (1990)
1950	pack aluminide coating	NiAl, CoAl	surface coating for protection from environment	Nicholls and Stephenson (1991)
1956	Kanthal Super, Mosilit	MoSi_2	electric heating elements	Fitzer and Rubisch (1958)
1961	A 15 compound	Nb_3Sn	superconductors	Westbrook (1977), Geballe and Hulm (1986)
1962	Nitinol	NiTi	shape memory alloy	Delacy et al. (1974) Hodgson (1990)
1967	Co-Sm magnets	Co_5Sm	permanent magnets	Stadelmaier et al. (1991), Westbrook (1977),

Examples of interesting intermetallics

for structural and functional applications

shape memory alloys

NiTi (B2)

Ni-based superalloys

(Ni)+Ni₃Al

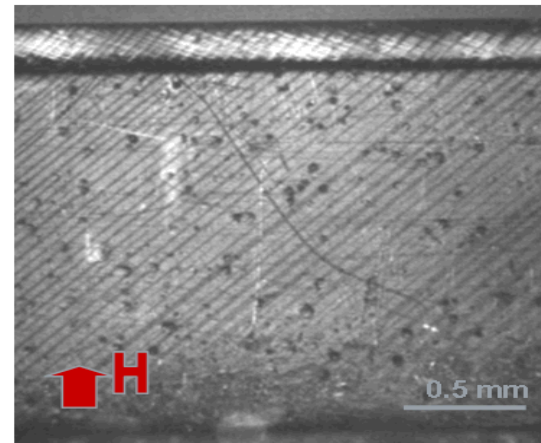
permanent magnets

Nd₂Fe₁₄B

ferromagnetic shape memory alloys

Ni₂MnGa (Heusler)

MSM  www.MagneticShape.de



© Ryan Lai & Jeffrey McCord, IFW Dresden

Source: www.magneticshape.de/

Intermetallic compounds: Issues of plastic deformation

Problem: brittleness

Plastic deformation:

Usually more difficult than in pure metals
due to more complex crystal structures

From experience:

brittleness increases with decreasing lattice symmetry and
increasing unit cell size

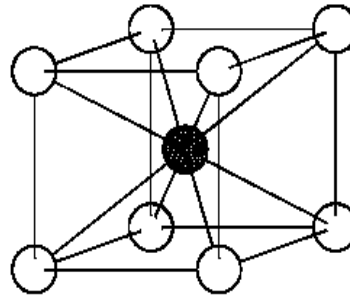
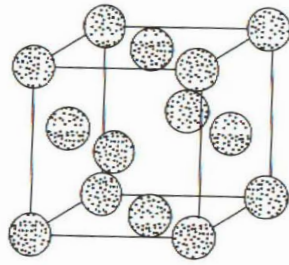
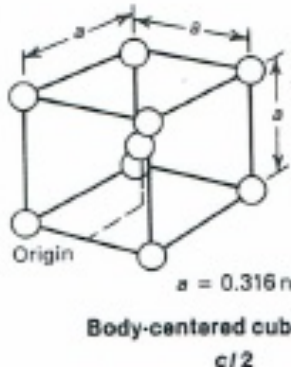
Therefore, intermetallics with high crystal symmetry and
small unit cells are preferred for developing structural materials

Crystal structures: “Cubic” systems

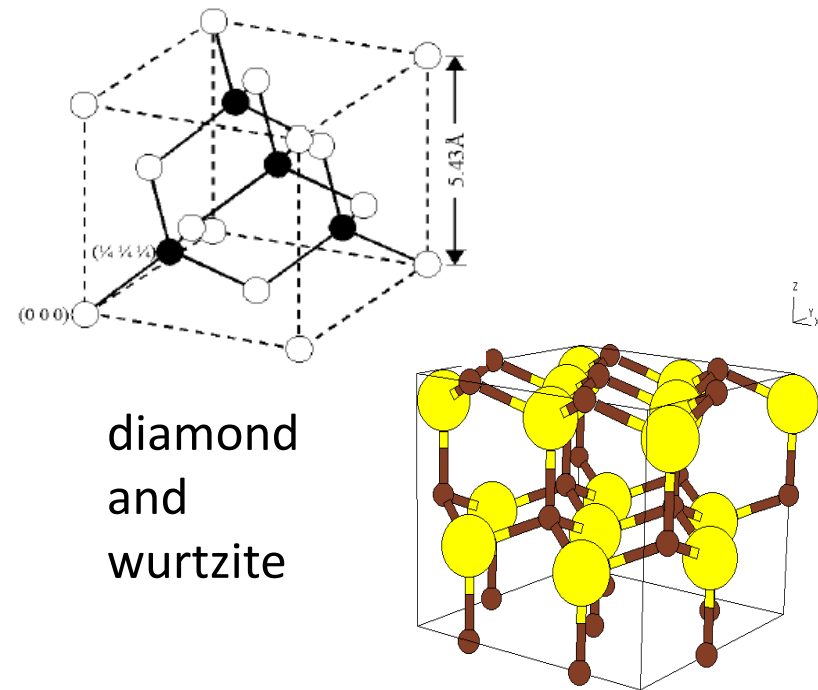
from fcc, bcc to

B2, diamond, Si, wurtzite, Heusler, Perovskite phases

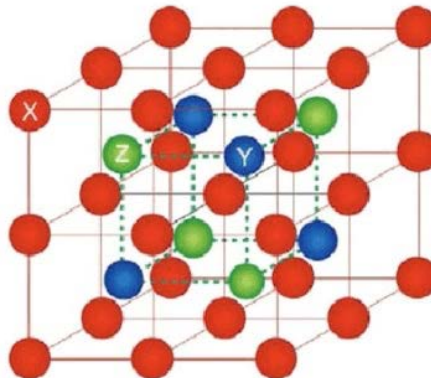
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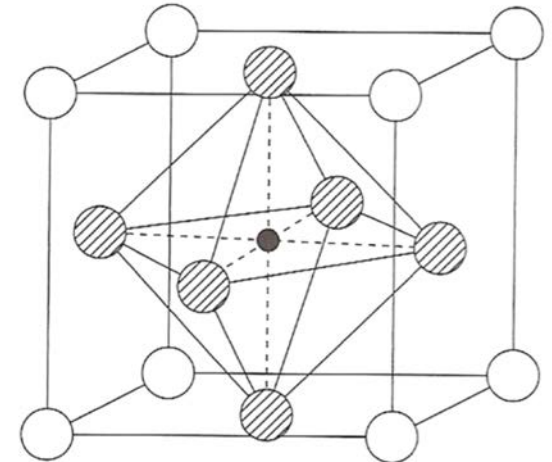
B2



$L2_1$ -type X_2YZ



Perovskite,
 ABO_3



Crystal structures: derivations

42 | 2 Crystal Structure and Chemical Bonding

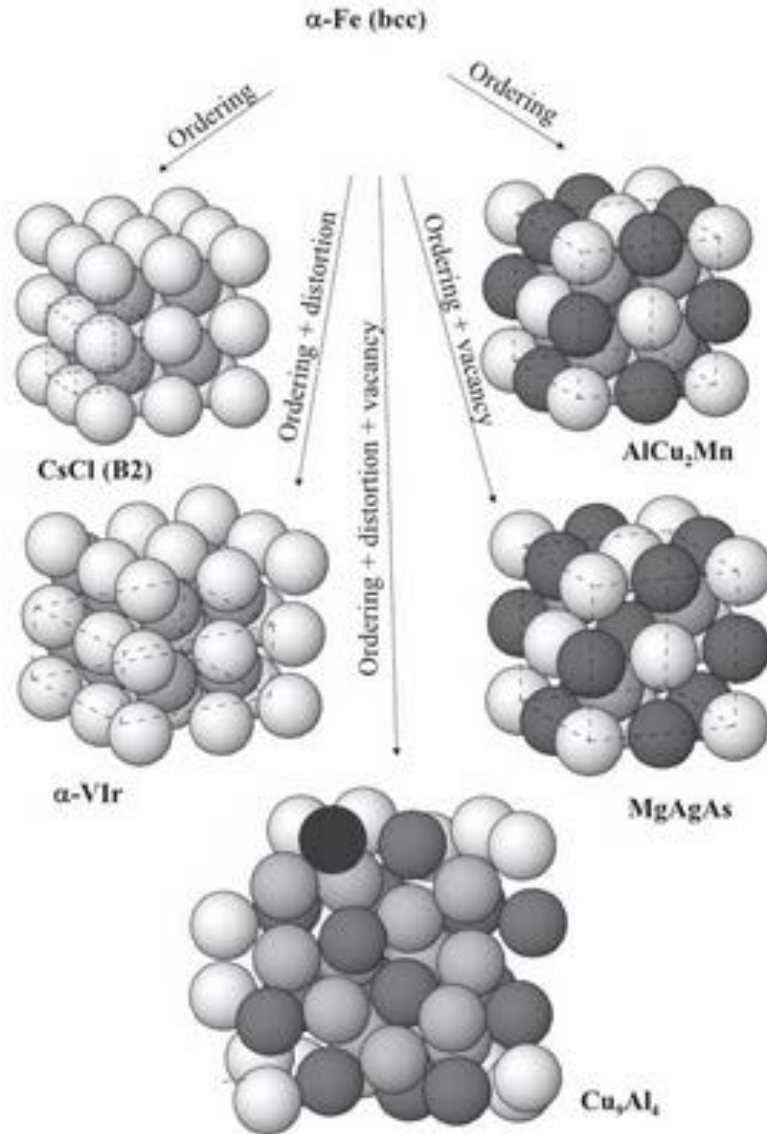
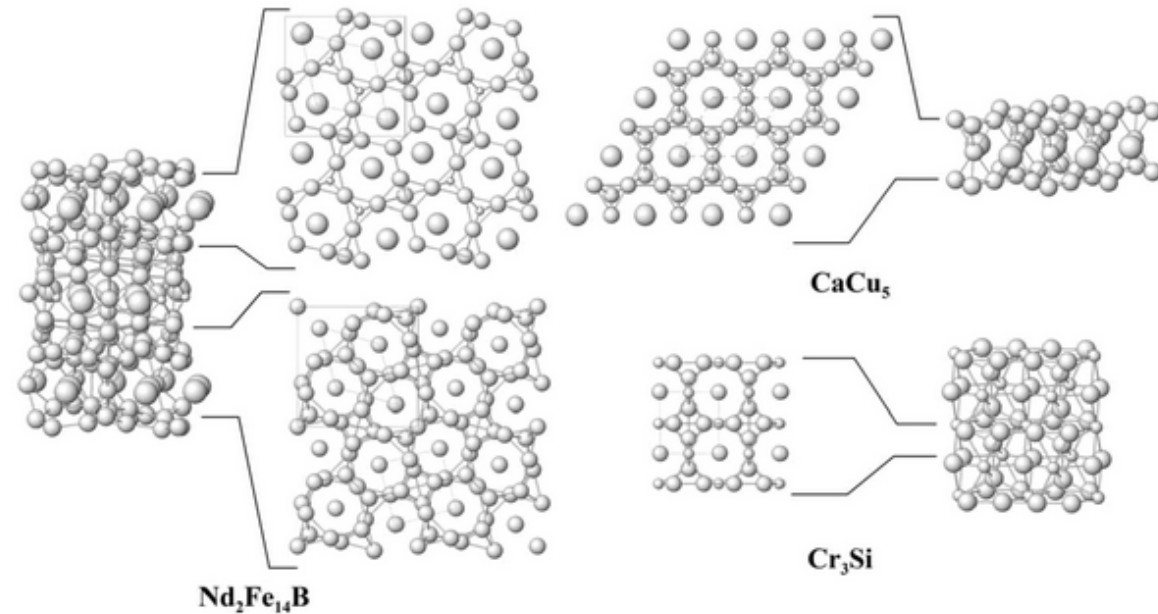


Fig. 2.16 Formation of structural patterns derived from the bcc packing.

crystal structure motifs


 Fig. 2.21 The crystal structure of the hard magnetic material Nd₂Fe₁₄B as a derivative of the crystal structure motifs of CaCu₅ and Cr₃Si.

Names of phases:

Mineral names, trivial names

Mineral names:

diamond, zinc blende, wurtzite,
quartz, rutil, pyrite, ilmenite, ...

naming after prototype structures

4.4 Intermetallische Phasen 147

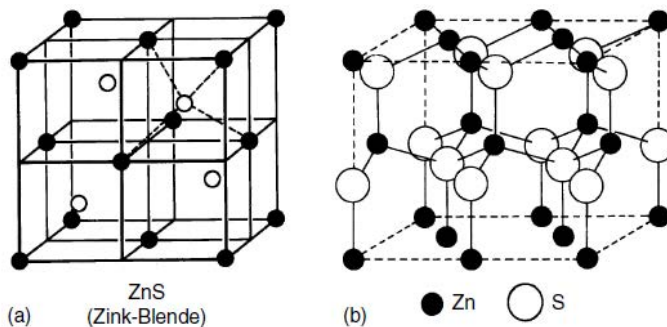


Abbildung 4.45. (a) Elementarzelle der Zinkblende-Struktur; (b) Elementarzelle des Wurtzitgitters.

- γ : γ -brass type or similar structures
- ε : Mg type
- ζ : Mg type
- η : W_3Fe_3C or Ti_2Ni type
- μ : W_6Fe_7 type
- σ : σ phase or σ -CrFe type
- χ : α -Mn or Ti_5Re_{24} type
- ω : ω_2 -(Cr, Ti) type (similar to the AlB_2 type)
- E : $PbCl_2$ or $CoSi_2$ type
- G : G phase, Th_6Mn_{23} or $Cu_{16}Mg_6Si_7$
- P : P phase or P-(Cr, Mo, Ni)
- R : R phase or R-(Co, Cr, Mo)
- T_1 : W_5Si_3 type
- T_2 : Cr_5B_3 type

rt: room-temperature modification
ht: high-temperature modification
lt: low-temperature modification

Names from Strukturbericht

letter, number, number

<u>Strukturbericht</u>	<u>Type</u>	<u>Examples</u>
A	unaries	A1: cF4-Cu A2: cI2-W
B	binaries AB	
C	binaries AB ₂	B2: cP2-CsCl
D	A _m B _n compounds	
E, F, ..., K	complex compounds	L1 ₀ : tP2-AuCu L1 ₂ : cP4-AuCu ₃
L	alloys	
O	organic compounds	
S	Silicates	

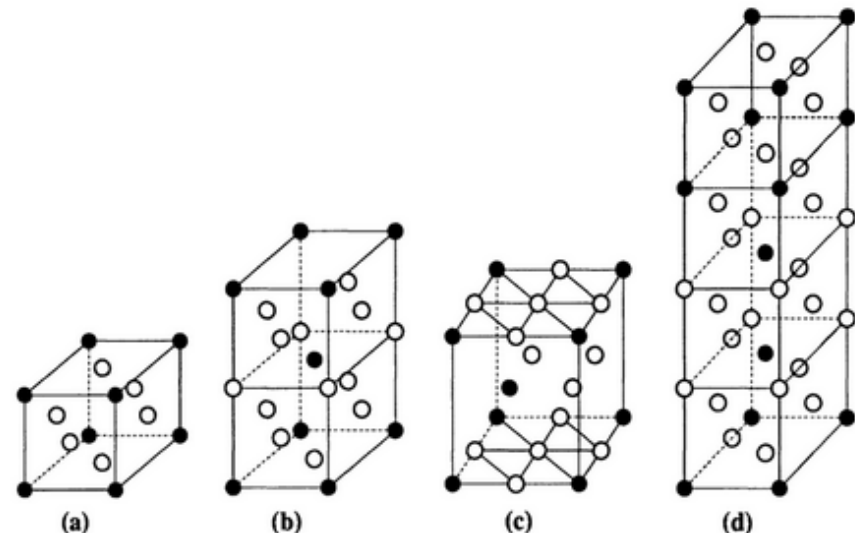


Fig.1 The unit cell of (a) L1₂, (b) D0₂₂, (c) D0₁₉ and (d) D0₂₃ structures.

Intermetallic phases named by scientists

Frank-Kasper phases

- Laves phases, e.g. hP12-MgZn₂, cF24-Cu₂Mg, hP24-Ni₂Mg
- σ phases, e.g. tP30
- other phases

Hume-Rothery phases

- electron compounds, VEC (valence electron concentration), same structural types for same VEC in well-defined ranges, e.g. in brass (Cu-Zn)

Heusler phases

- cF16-MnCu₂Al (a derivative of the CsCl-type)

Zintl phases

- polar, salt-like phases, e.g. Zintl boundary in Periodic Table (alkali-, alkaline elements)+main group (>4) elements

Hägg phases

- interstitial phases based on occupancy of interstices in close-packed structure of transition metals by small non-metals (H,B,C,N)

Nowotny phases, Chevrel phases, ...

also: martensite, austenite, ...

Descriptors for intermetallic phases

- chemical composition and existence range (p, T)
- structure type
(crystal system, space group, number of atoms per unit cell, list of occupied atomic positions)

non-stoichiometric phases:
formula denotes: „ideal composition“

variable composition,
e.g. for solid solution (Ni, Cu) or $\text{Ni}_x\text{Cu}_{1-x}$
(0...1)

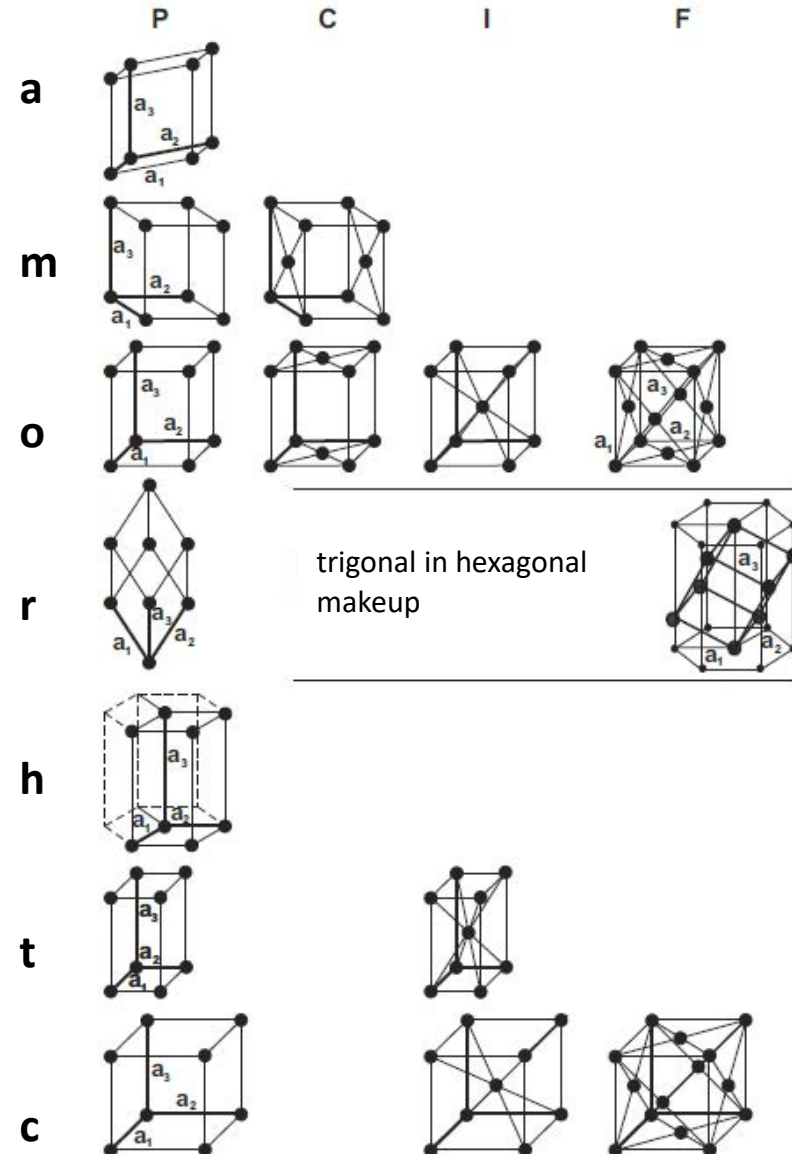
Crystal systems and Bravais lattices

Single crystals: chemically and structurally homogeneous materials with anisotropic properties

7 crystal systems:

(fundamental point lattices from crystal morphology)

- triclinic (anorthic) a
- monoclinic m
- orthorhombic o
- trigonal (rhombohedral) r
- hexagonal h
- tetragonal t
- cubic c



14 Bravais lattices:

7 from centering translations, i.e. lattice points in interior of primitive cell

- primitive P
- side centered C
- body centered I
- face centered F

Crystal structures: Pearson symbols

consist of:

system symbol (small), lattice symbol (capital), number of atoms in unit cell

system symbol

- triclinic (anorthic) **a**
- monoclinic **m**
- orthorhombic **o**
- tetragonal **t**
- hexagonal
(trigonal, rhombohedral) **h**
- cubic **c**

lattice symbol

- P** primitive
- I** body centered
- F** all face centered
- C** side face centered
- R** rhombohedral

e.g.:

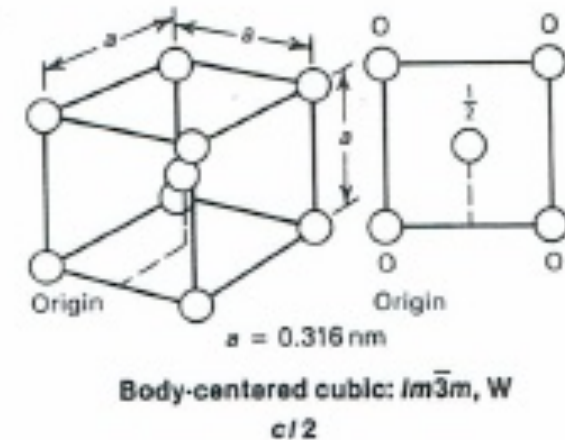
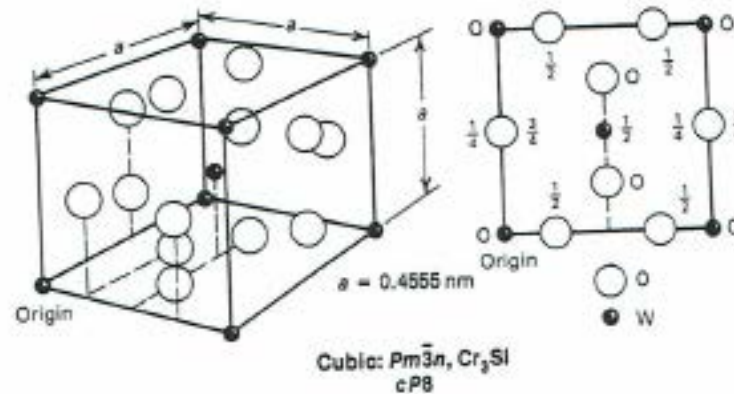
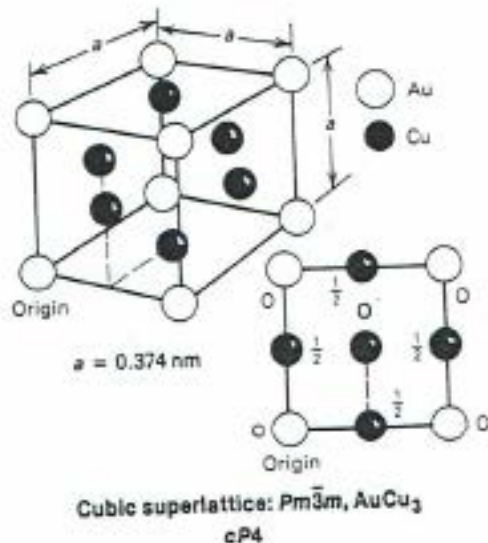
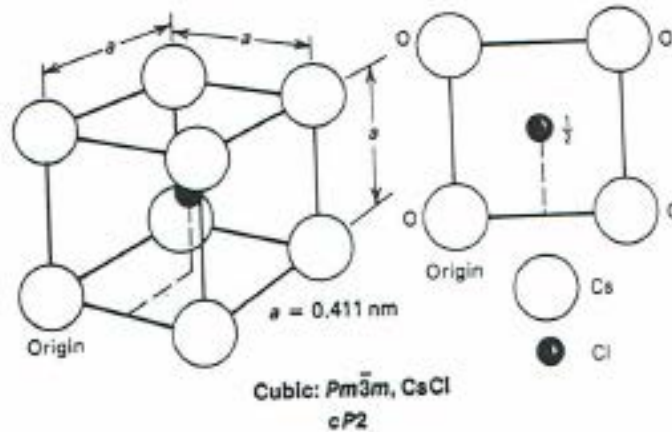
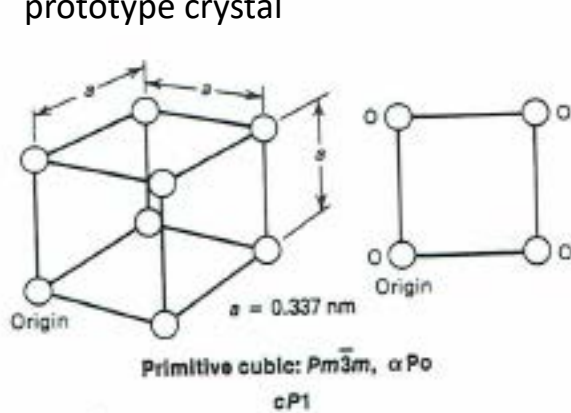
cl2 (α -Fe)

tP2 (L1₀ FePt)

Examples of cubic crystal structures

Schematic drawings of the unit cells and atom positions for some metal crystals, Pearson symbol, space lattice, crystal system, space-group notation, prototype, lattice parameters of prototype crystal

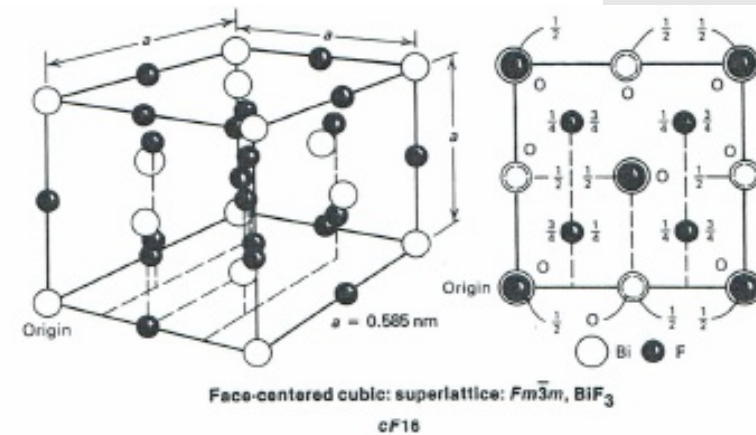
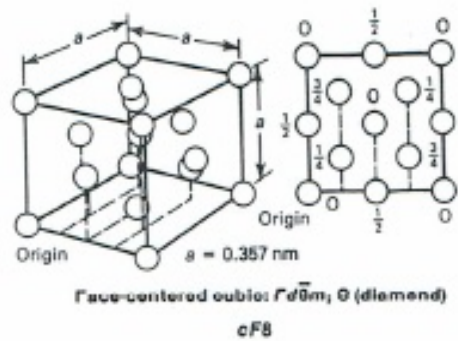
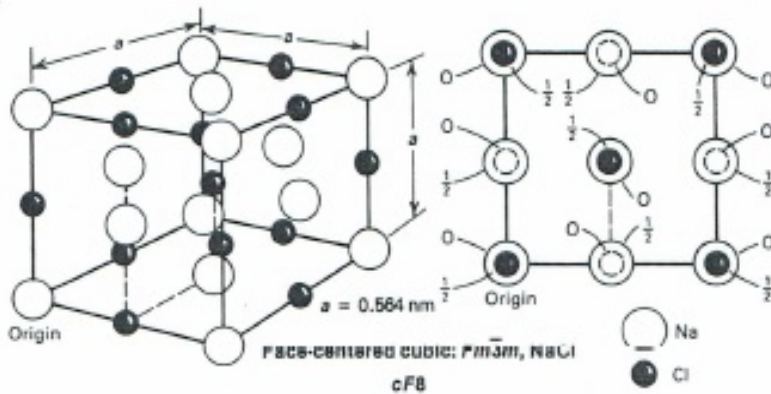
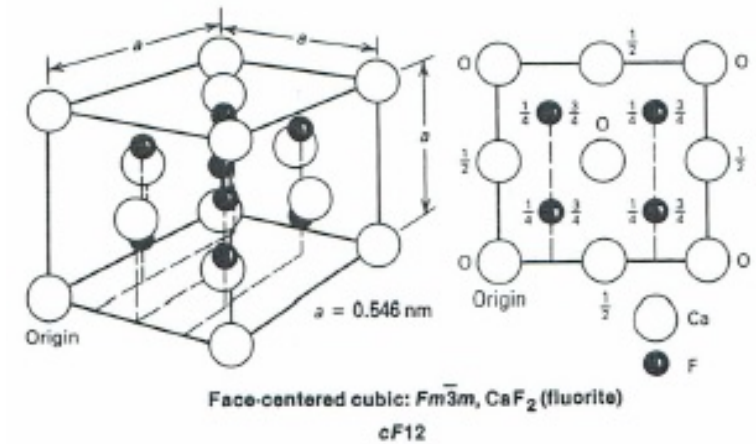
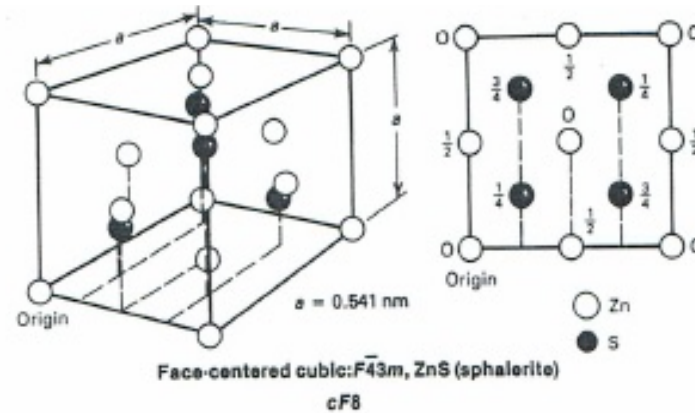
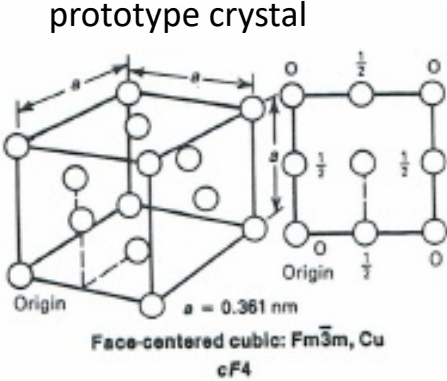
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Face-centered cubic crystal structures

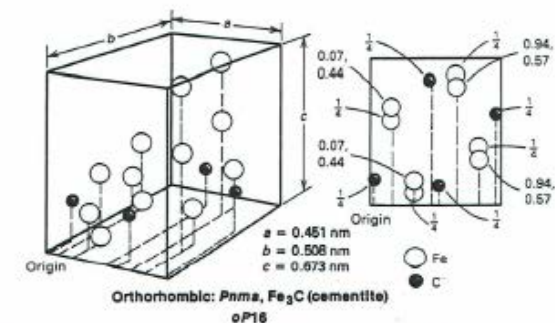
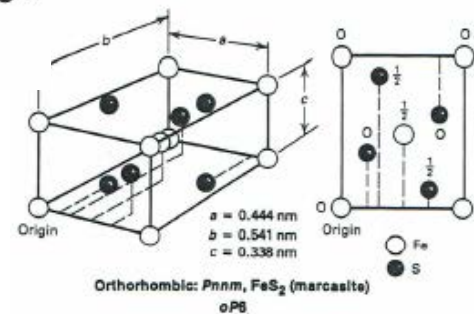
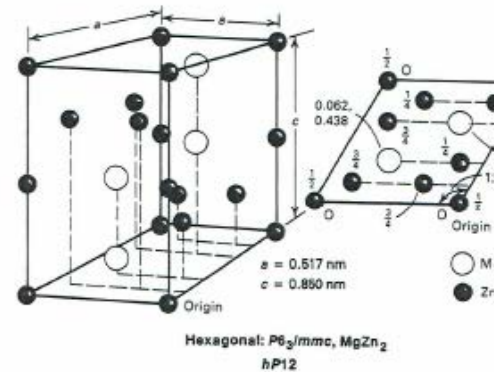
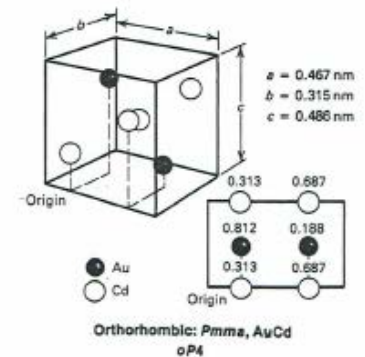
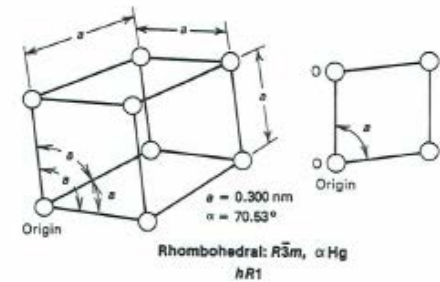
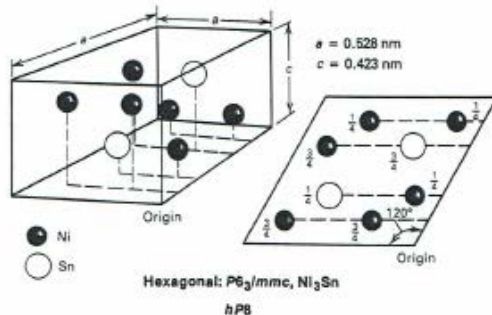
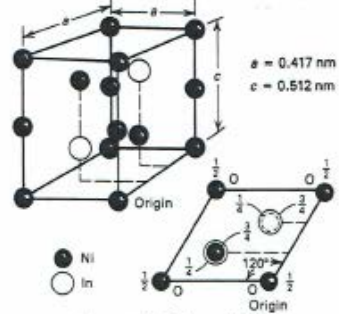
Schematic drawings of the unit cells and atom positions for some metal crystals, Pearson symbol, space lattice, crystal system, space-group notation, prototype, lattice parameters of prototype crystal

RUB



Examples of intermetallic crystal structures

Schematic drawings of the unit cells and atom positions for some metal crystals,
Pearson symbol, space lattice, crystal system, space-group notation, prototype, lattice parameters of
prototype crystal



Crystal structures: point groups

32 crystallographic point groups

point symmetry operations

- identity operation 1
- rotations 2, 3, 4, 6 (-fold)
polar and non-polar rotation axes

proper rotations N

(transform an object congruently)

rotations through angles $\frac{2\pi}{N}$

improper rotations $\bar{N}$

rotoinversion axis $\bar{1}$
 mirror plane $\bar{2} = m$
 $\bar{3}$
 $\bar{4}$
 $\bar{6} = 3/m$

- at least 1 point invariant
- 32 point groups are compatible with 3D point lattices

32 crystallographic point groups

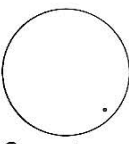
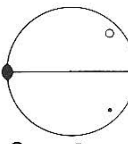
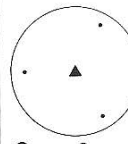
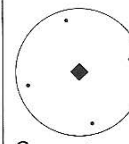
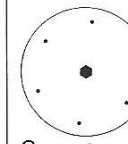
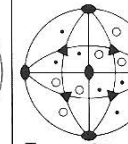
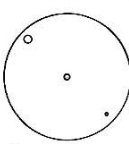
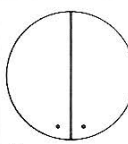
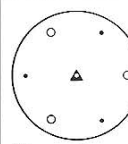
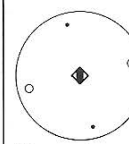
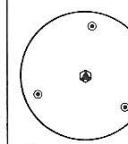
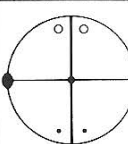
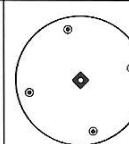
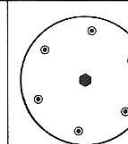
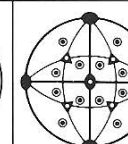
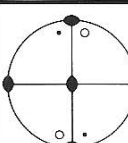
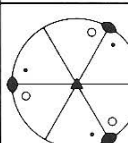
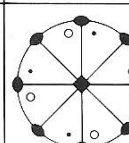
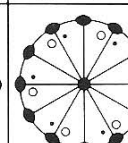
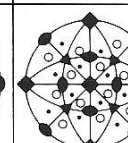
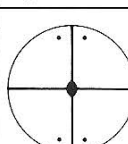
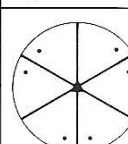
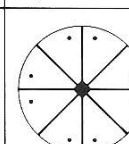
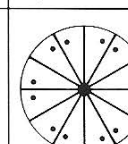
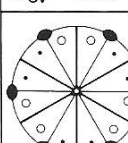
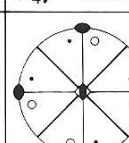
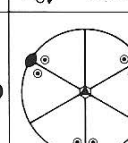
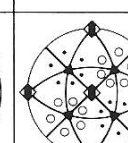
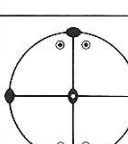
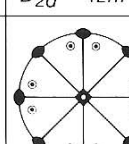
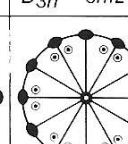
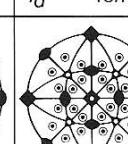
Triclinic	Monoclinic / Orthorhombic	Trigonal	Tetragonal	Hexagonal	Cubic
 C_1 1	 C_2 2	 C_3 3	 C_4 4	 C_6 6	 T 23
 C_i $\bar{1}$	 C_s m	 C_{3i} $\bar{3}$	 S_4 $\bar{4}$	 C_{3h} $\bar{6}$	
	 C_{2h} 2/m		 C_{4h} 4/m	 C_{6h} 6/m	 T_h m $\bar{3}$
	 D_2 222	 D_3 321	 D_4 422	 D_6 622	 O 432
	 C_{2v} mm2	 C_{3v} 3m1	 C_{4v} 4mm	 C_{6v} 6mm	
		 D_{3d} 3m1	 D_{2d} 42m	 D_{3h} 6m2	 T_d 43m
	 D_{2h} mmm		 D_{4h} 4/mmm	 D_{6h} 6/mmm	 O_h m3m

Figure 1-30. Stereographic projections of the 32 crystallographic point groups arranged in columns according to which of the 7 crystal systems they belong to. The order of the point groups increases from top to bottom. Both the Schoenflies and the Hermann–Mauguin (international) symbols are given. Rotation axes N are marked by black graphical symbols with N -fold rotation symmetry, rotoinversion axes by a white and black symbol with N -fold rotation symmetry, the inversion center by an open circle or a white dot when sited on a rotation axis; in-plane mirror planes are indicated by bold limiting circles on the stereographic projection, when normal to the plane they are marked by bold lines. Polar rotation axes parallel to the projection plane are marked by one graphical symbol only. The action of the symmetry operations on a pole (filled circle when above the paper plane, open circle when below) is shown also.

Schoenflies:

C_n :	cyclic	$n=1,2,3,4,6$
D_n :	dihedral	s, v, h : mirror planes
T :	tetrahedral	i : inversion center
O :	octahedral	

International (Herrmann-Mauguin):

e.g.: 2/m („two over m“)
= a twofold rotation axis
is perpendicular to a mirror plane

Crystal structures: 230 space groups

space groups

describe **spatial symmetry**
of a crystal on an atomic level

230 space groups

all regular crystal structures belong to one
of the 230 space groups

space groups are associated with:

- a given point group
- a Bravais space lattice
- the action of the point group on the lattice

- screw axes
- glide planes

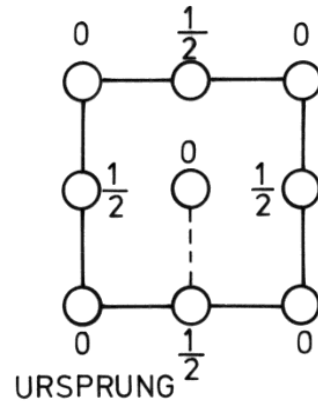
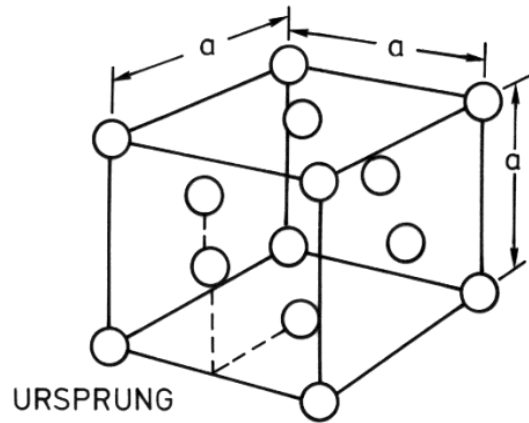
Wyckoff positions (Punktlage):

“a point belonging to a set of
points for which site symmetry
groups are conjugate subgroups
of the space group”

each atom of a crystal structure can
be assigned to a particular Wyckoff
position with a given site symmetry

Order and disorder in crystal structures

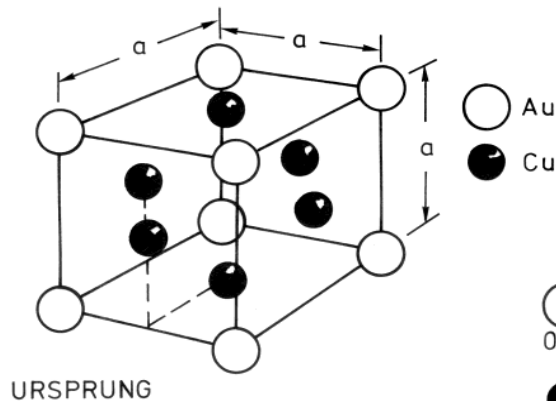
face centered cubic (fcc), Cu-type, A1, cF4



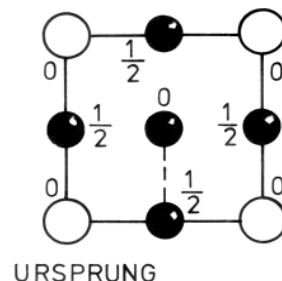
e.g.: Cu, Ni, γ -Fe, Al, ...

Cu - Typ

ordered fcc, L1₂, cP4



in ordered crystals,
specific atoms occupy
specific lattice sites

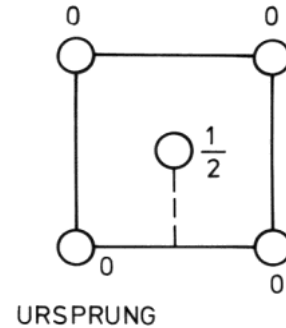
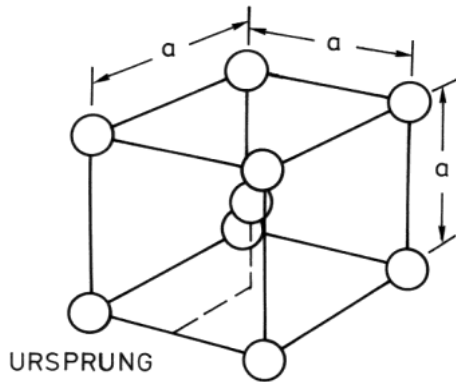


technologically important:
Ni₃Al

L1₂
AuCu₃-Typ

Order and disorder in crystal structures

body centered cubic (bcc), W-type, A2, cI2

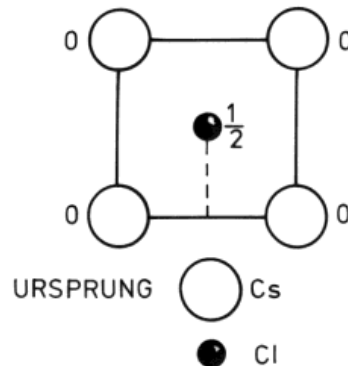
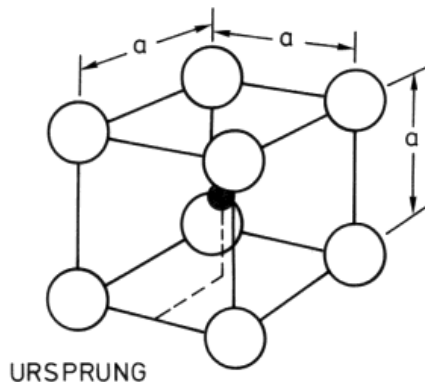


W- Typ

e.g.: W, α -Fe,
 δ -Fe, Mo, V, Cr

ordered bcc (B2, CsCl-type, cP2)

e.g.: NiTi, NiAl,



Cs Cl - Typ

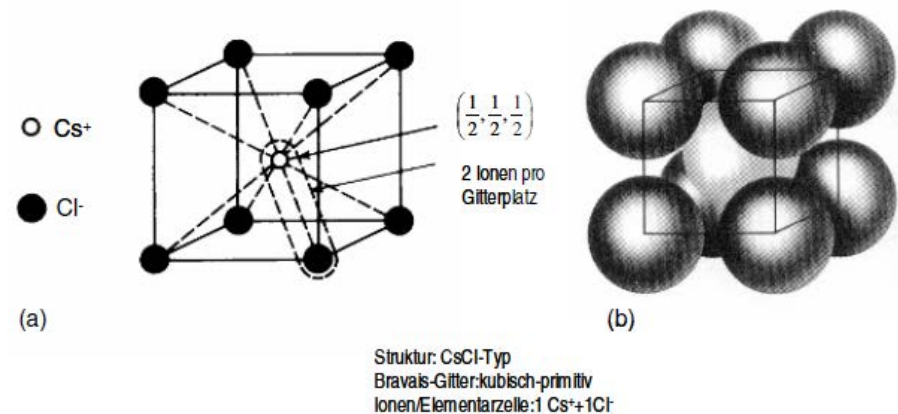
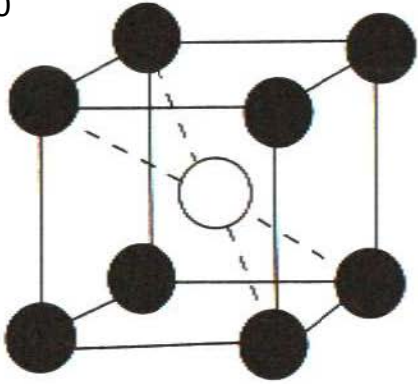


Abbildung 2.20. Elementarzelle von Cäsiumchlorid (CsCl); (a) Lage der Atome im Gitter; (b) Modell der harten Kugeln.

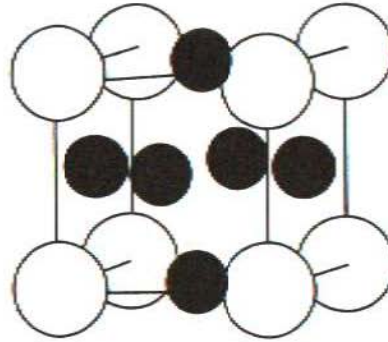
Ordered crystal structures

$L2_0$



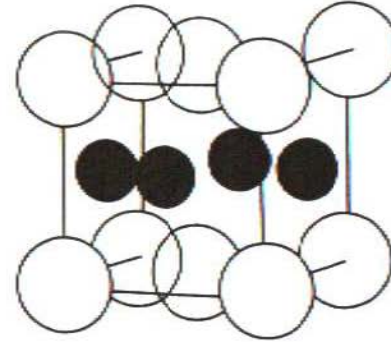
(a) ● Cu ○ Zn

$L1_2$



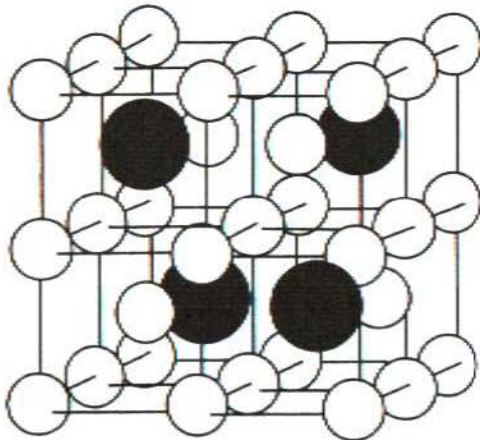
(b) ● Cu ○ Au

$L1_0$



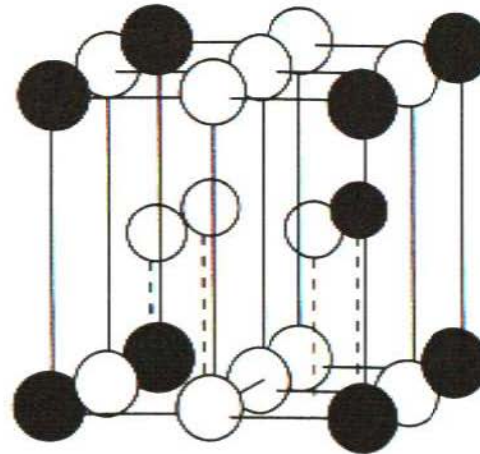
(c) ● Cu ○ Au

$D0_3$



(d) ● Al ○ Fe

$D0_{19}$



(e) ● Cd ○ Mg

Order and disorder in alloys

equiatomic AB alloy

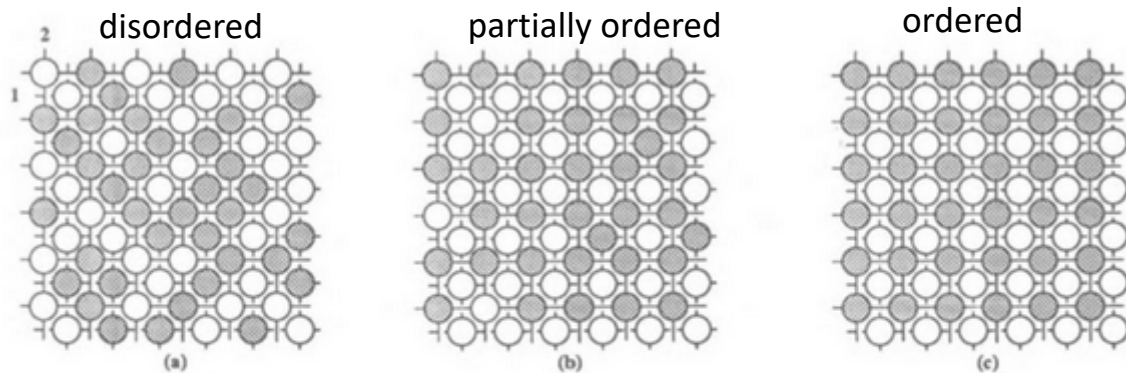


Figure 1: Two dimensional depiction of an equiatomic AB alloy. We have identified two sublattices (1,2) and two atomic species ($A \equiv$ unshaded; $B \equiv$ shaded), (a) disordered $A_{0.5}B_{0.5}$ solid solution for which the long range order parameter $\eta = 0$, (b) a partially ordered alloy $0 \leq \eta \leq 1$ (c) ordered AB intermetallic compound $\eta = 1$.

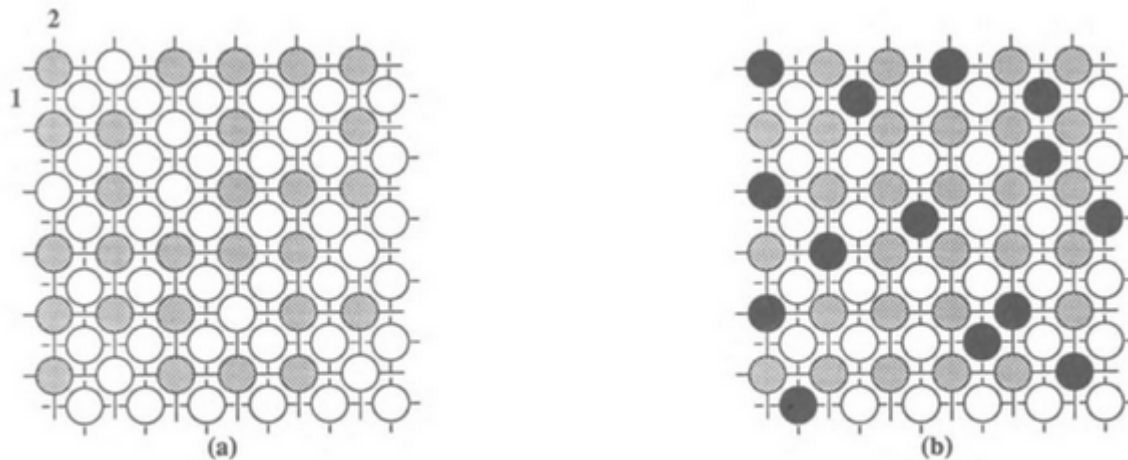
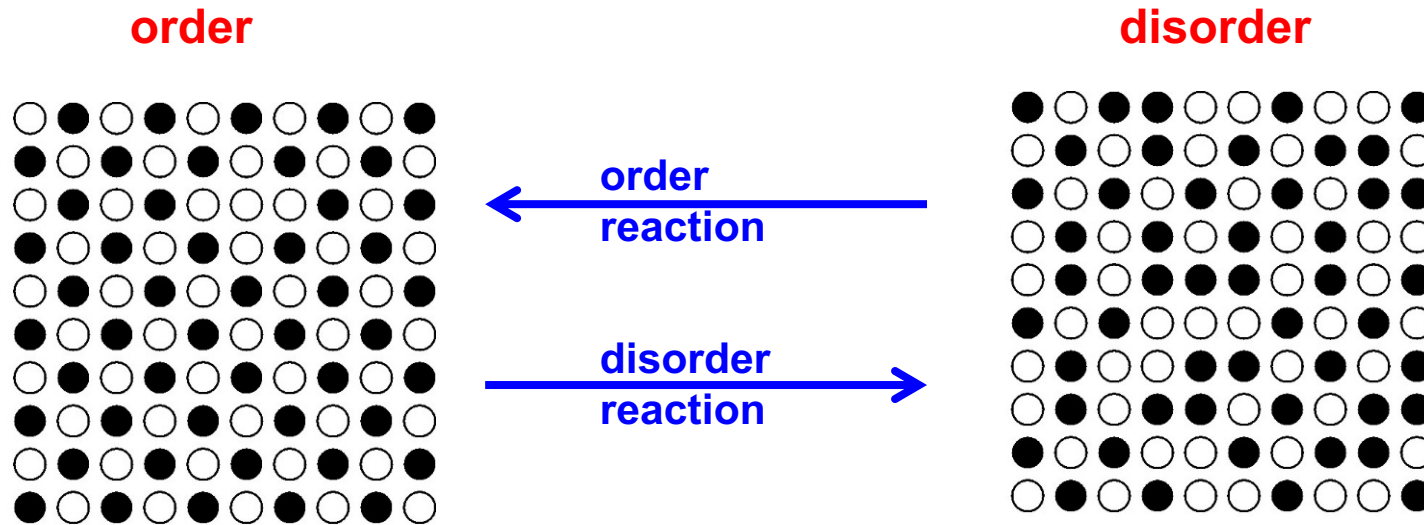


Figure 2: Two dimensional depiction of (a) off-stoichiometric AB ordered intermetallic, (b) ternary addition to binary AB ordered intermetallic

Order and disorder reactions



e.g. in the systems
Cu-Au, Cu-Zn

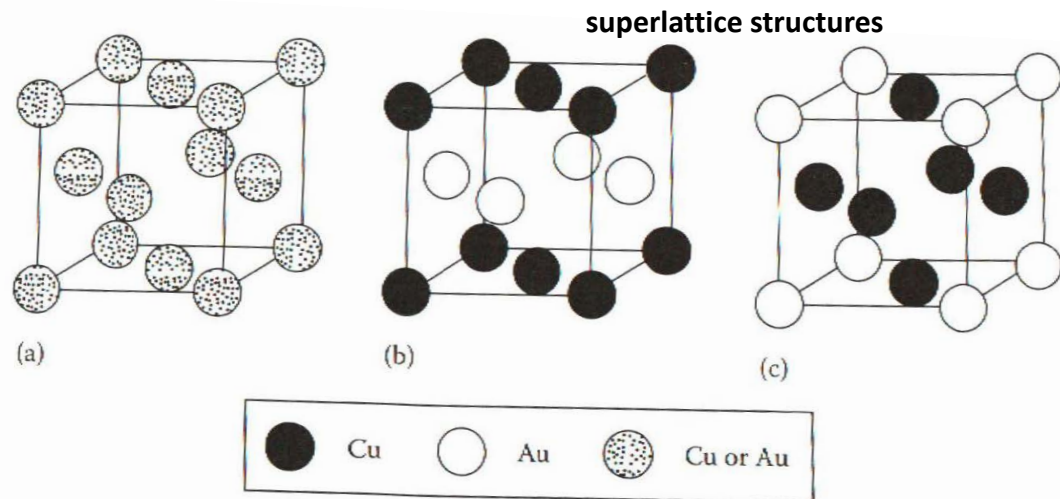
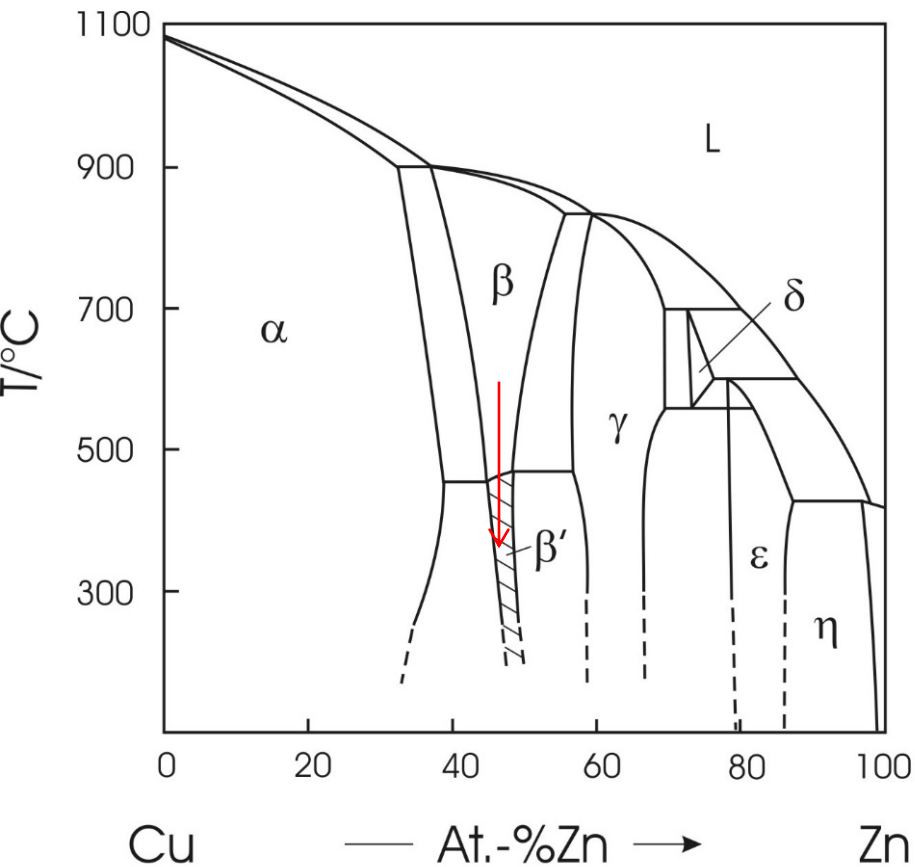


FIGURE 1.20

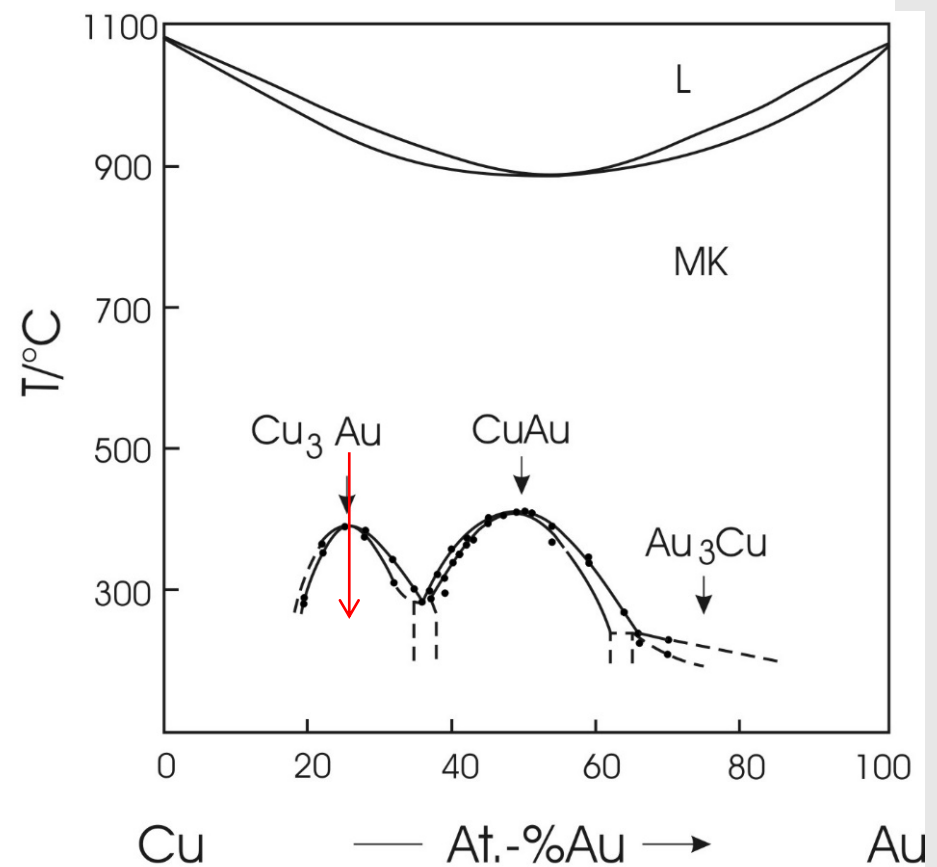
Ordered substitutional structures in the Cu-Au system: (a) high-temperature disordered structure, (b) CuAu superlattice, (c) Cu₃Au superlattice.

Order and disorder reactions in phase diagrams



order reaction:

$\beta \rightarrow \beta'$



order reaction:

solid solution $\rightarrow \text{Cu}_3\text{Au}$

Quantifying order

Order parameter p (Bragg/Williams):

system: AB

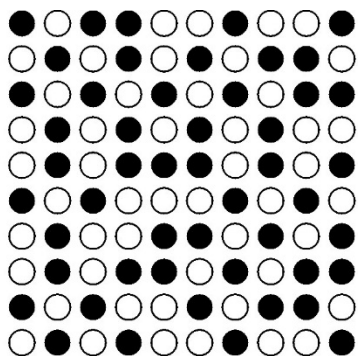
fully ordered state: all A-atoms on α -sites, all B-atoms on β -sites

q : fraction of A-atoms on α -sites

$1-q$: fraction of A-atoms on β -sites

$$p = 2 \cdot q - 1$$

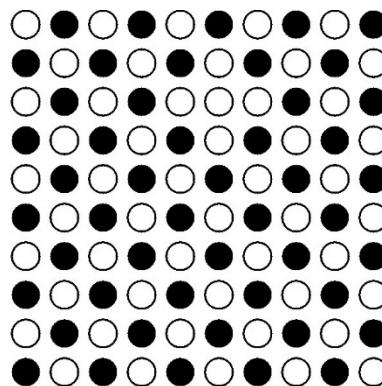
fully disordered



$$q=0.5$$

$$p=0$$

fully ordered



$$q=1$$

$$p=1$$

Quantifying order

intermetallic compound A_mB_n

i.e. m atoms of A and n atoms of B

p_α – fractions of A atoms on A-places,

p_β – fraction of B atoms on B-places,

$x_A = m/(m+n)$,

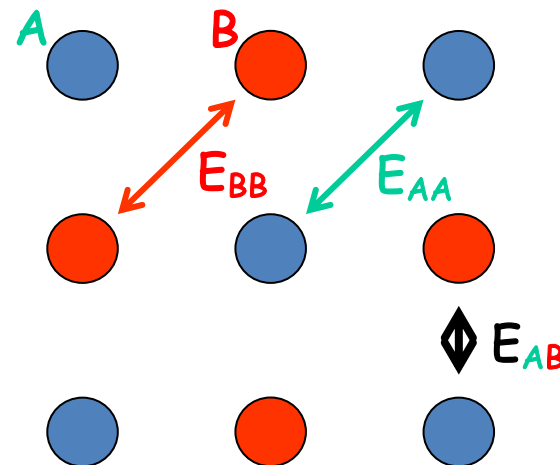
$x_B = n/(m+n)$

order parameter:

$$p = \frac{p_\alpha - x_A}{1 - x_A} = \frac{p_\beta - x_B}{1 - x_B}$$

For order/disorder reactions the interaction W between atoms plays an important role:

$$W = E_{AB} - 1/2 \cdot (E_{AA} + E_{BB})$$



Cu-Au

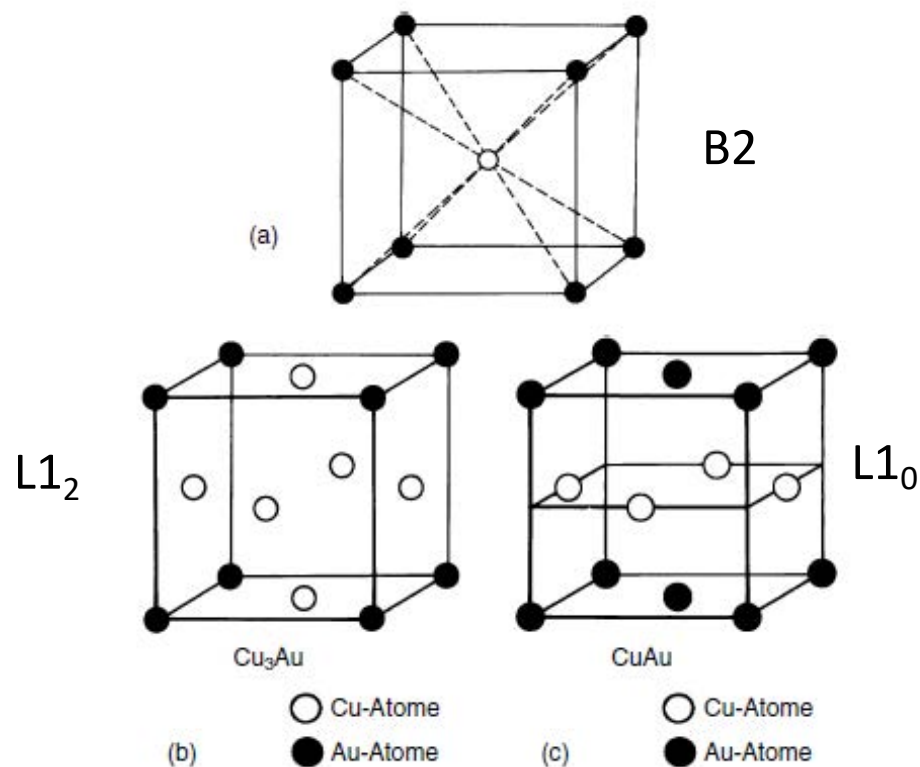


Abbildung 4.33. Geordnete Atomverteilungen vom Typ AB lassen sich im CsCl-Gitter (a) (B2-Struktur), solche vom Typ AB_3 im Cu_3Au -Gitter (b) (L1_2 -Struktur) verwirklichen, die mit einem krz bzw. kfz Mischkristall bei regelloser Atomverteilung verträglich sind. Ein kubisch flächenzentrierter Mischkristall vom Typ AB (c) kann nicht ordnen ohne seine kubische Struktur zu verlieren. Bei einer Unterteilung des kfz Gitters in je zwei Untergitter für A und B kommt es zur Schichtenbildung und damit wegen der unterschiedlichen Atomradien A und B zu einer tetragonalen Kristallstruktur.

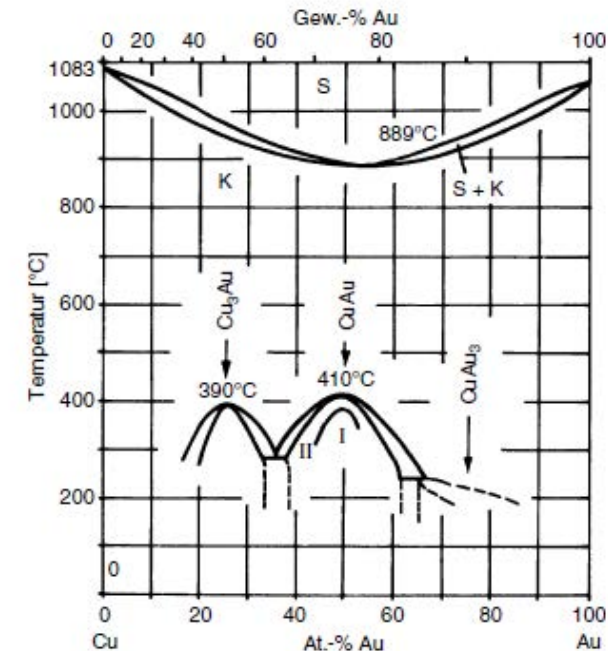


Abbildung 4.34. Zustandsdiagramm des Systems Cu-Au, das bei tieferen Temperaturen verschiedene geordnete Phasen im festen Zustand bildet, die aber bereits weit unterhalb des Schmelzpunktes wieder in eine regellose Verteilung übergehen [4.1].

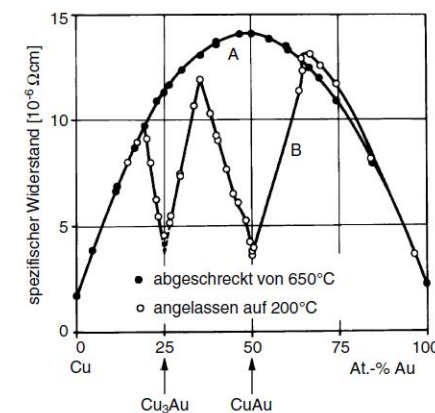


Abbildung 4.39. Das Auftreten von Überstrukturen führt zur starken Verringerung des elektrischen Widerstandes, bspw. für Cu_3Au und CuAu (nach [4.6]).

Temperature-dependence of order

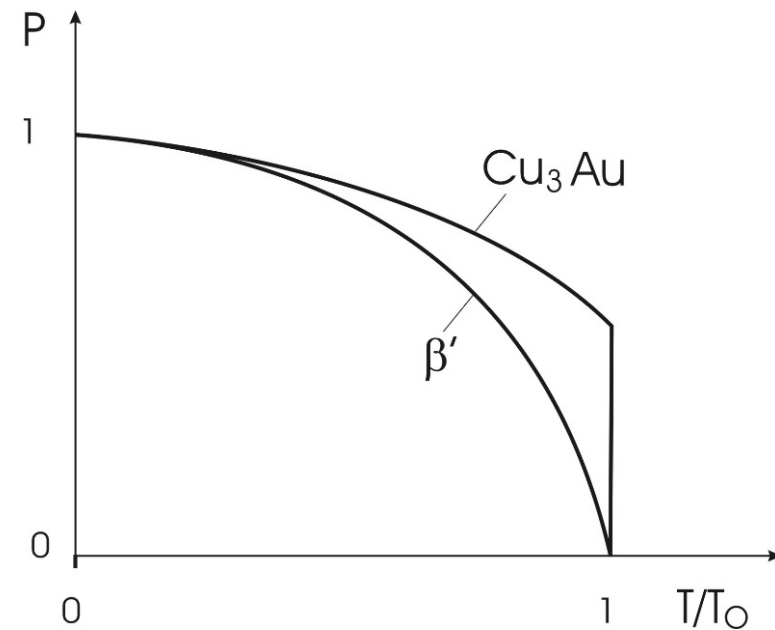
(1) critical temperature T_0

$T < T_0$ system is ordered

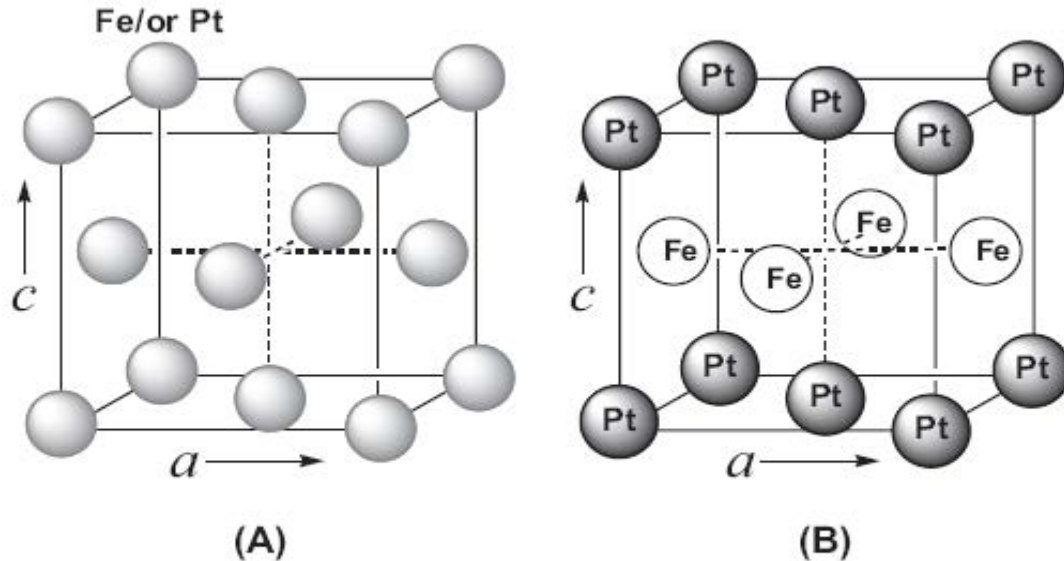
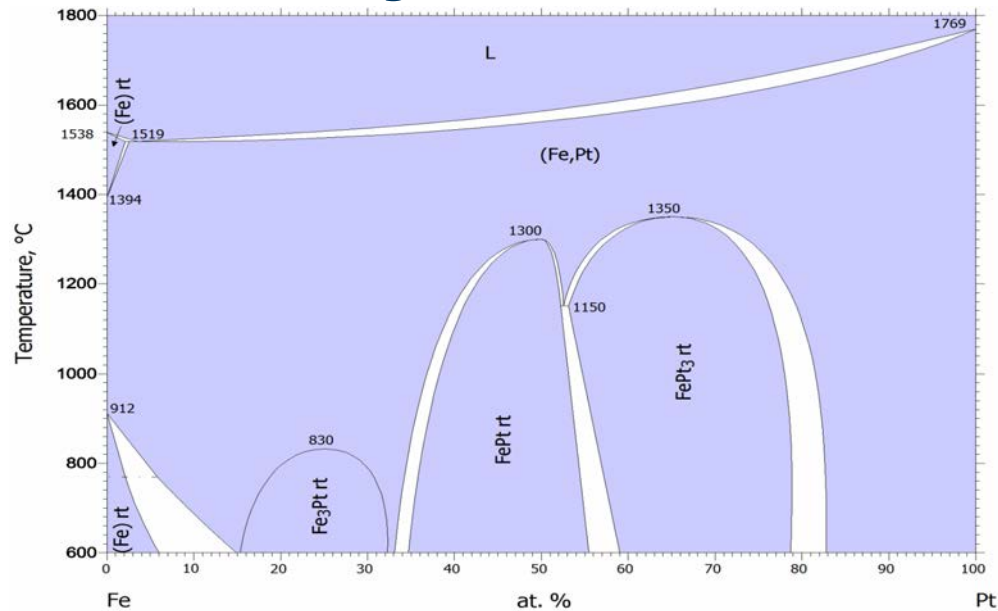
$T > T_0$ system is not ordered

(2) order parameter p is temperature-dependent

- up to $0.3 T_0$ order is maintained
- approaching T_0 , order decreases
- T_0 is proportional to interaction energy W



The Fe-Pt system



$L1_0$
unique
magnetic
properties

Figure 2. Schematic illustration of the unit cell of A) chemically disordered fcc and B) chemically ordered fct FePt.

Rules for formation of intermetallic phases

Intermediate and intermetallic phases

for a lot of intermetallic phases *simple valence concepts are not applicable*, some compounds have a broad existence range, others not

Some examples of intermetallics:

- Valence compounds
 - NiAs-compounds
 - Hume-Rothery phases
 - Laves-phases (AB_2)
 - Zintl-phases
 - TCP phases: topologically close packed phases
- constituents A, B tend to form a compound, if properties are different (size, electronegativity, valence electron concentration, (electro)chemical properties, ...)
 - from supersaturated solid solutions over intermediate phases to stoichiometric compounds
 - binding can be a mixture of metallic, covalent, ionic binding; the higher the degree of covalent and ionic bonding, the higher is usually brittleness and resistivity (non-metallic properties)

Periodic tables of elements

Periods

increasing ionisation energy, increasing electronegativity

Groups: elements with similar chemical behavior:
similar outer electron configuration, same number of
valence electrons

The periodic table is color-coded by groups:

- Nichtmetalle
- Erdalkalimetalle
- Andere Metalle
- Actinoide
- Edelgase
- Alkalimetalle
- Übergangsmetalle
- Halogene
- Lanthanoidenreihe

Intermetallic phases

About 2/3 of elements are metals

Furthermore, alloys (based on metals): intermetallic phases (alloys, compounds) as well as metal-oxides, -nitrides, -carbides, hydrides, ...

Different groups of intermetallics

H																	He
Li	Be	metals										B	C	N	O	F	Ne
Na	Mg											Al	Si	P	S	Cl	Ar
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn

electropositive
metals

transition metals
(similar radii and electronegativity,
different number of valence electrons)

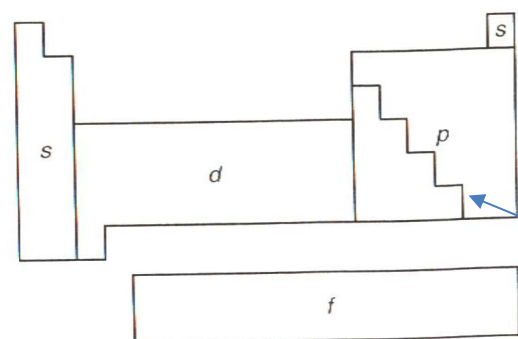
meta-
metals

semi-
metals

half-metals: act as conductor to electrons of one spin orientation, but as an insulator or semiconductor to those of the opposite orientation. Although all half-metals are ferromagnetic (or ferrimagnetic), most ferromagnets are not half-metals. Many of the known examples of half-metals are oxides, sulfides, or Heusler alloys.

very small overlap between bottom of conduction band and top of valence band. Semimetal has no band gap and a negligible density of states at the Fermi level.

Periodic tables of elements



valence electrons in orbitals
s, d, p, f

Zintl line between d (semi-metals) and p (non-metals)

Group number																	
4	5	6	7	8	9	10	11	12	13	14	15	16	17	18	1	2	
													H	He	Li	Be	
									B	C	N	O	F	Ne	Na	Mg	
									Al	Si	P	S	Cl	Ar	K	Ca	
Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr	Rb	Sr	
Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe	Cs	Ba	
Hf	Ta	W	Re	Os	Ir	Pt	Au										
Transition metal divide								Composite divide	Covalent divide				Ionic divide				

Figure 4.7. The metallurgist's Periodic Table showing the four 'divides' proposed by Stone (1979)

Notice that the position of the different groups has been shifted, in comparison with the conventional presentation of the Table, in order to give special emphasis to the ionic divide.

Table 2.1. The electronegative scale as reported by Pauling (1967).

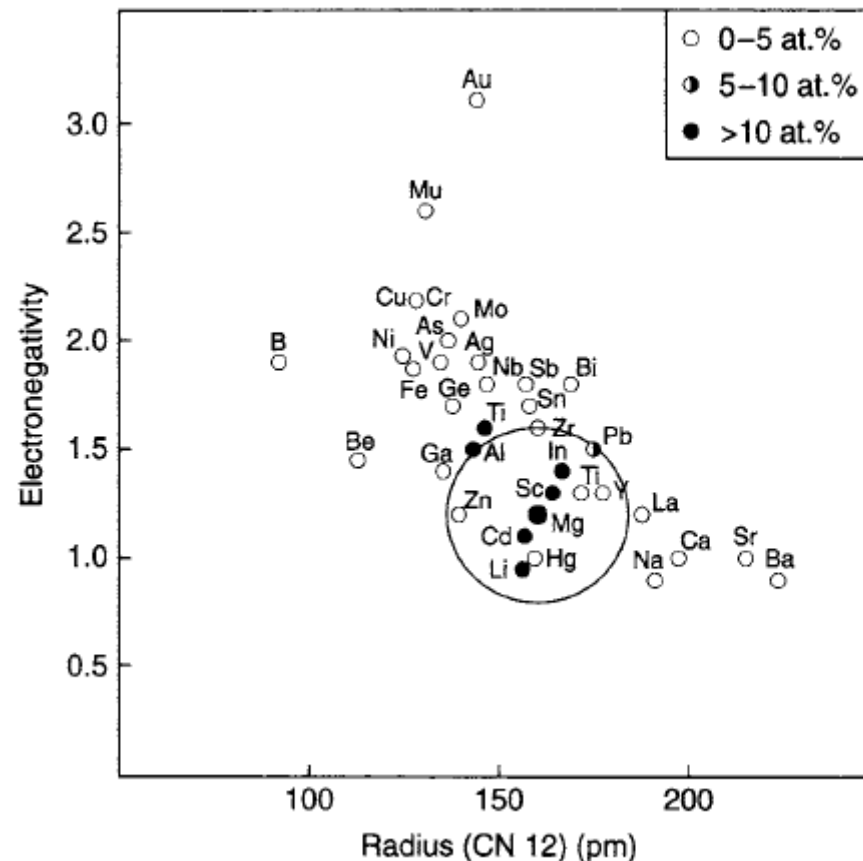
„the power of an atom in a material to attract electrons to itself“

Solubility: Darken and Gurry diagram

Darken and Gurry proposed rules for elemental solubility in 1953.

An ellipsoid is drawn around an element (e.g. Mg) showing the elements with a maximum variance of 15% atomic size and 0.3 electronegativity.

Elements inside the ellipsoid usually have a higher solubility in the respective element indicated by the color of the dots in the figure.



Mutual solubility in binary alloys of transition metals (X-Y)

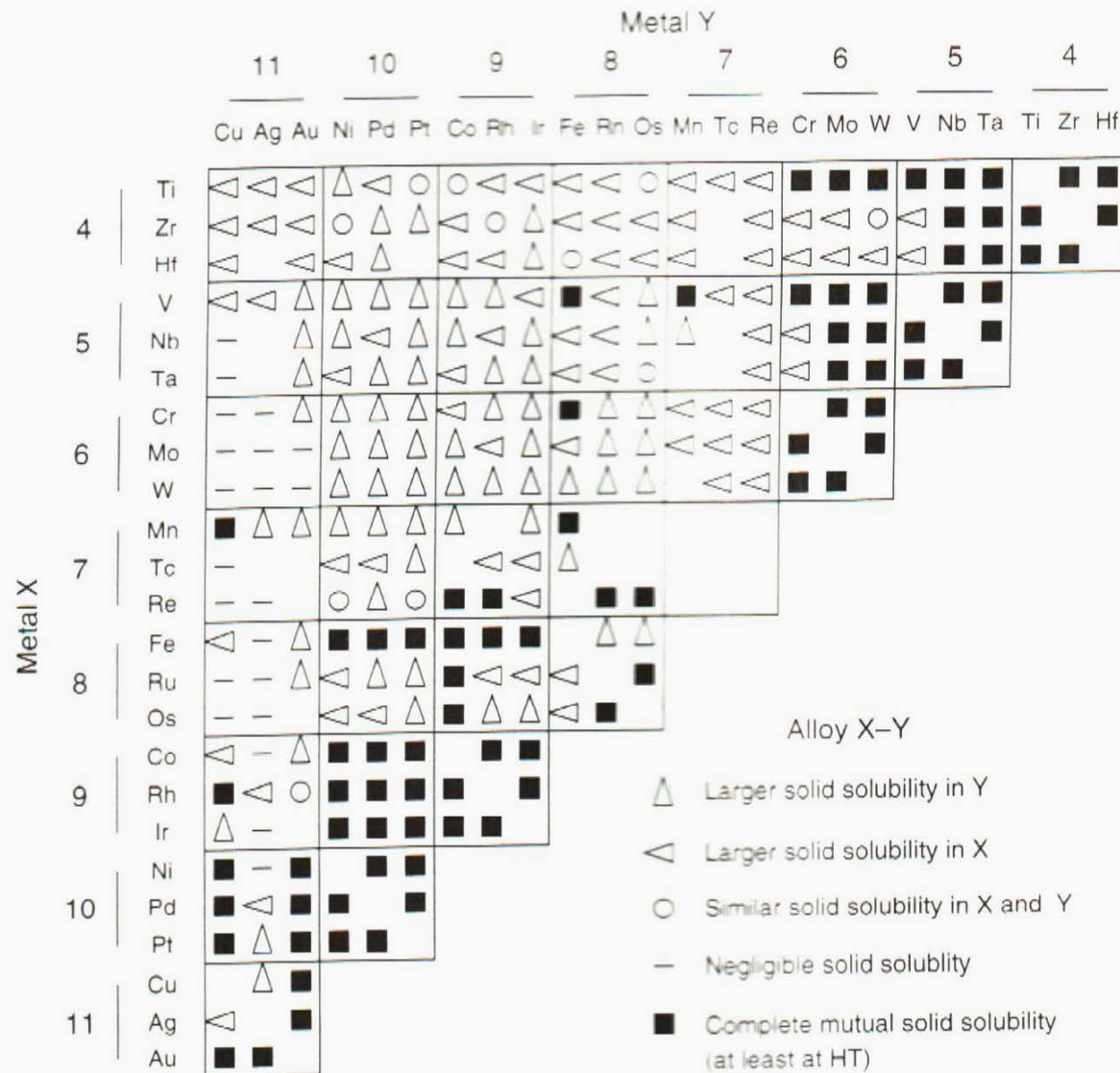


Figure 2.15. Relative extent of the mutual solid solubility in binary alloys of transition metals, ordered according to their group number in the Periodic Table. The group number is reported on the left and on the top of the figure.

Figure 1 consists of four subplots labeled (a) through (d), each showing a phase diagram of the order parameter T versus temperature T . The vertical axis is labeled T and the horizontal axis is labeled A to B . (a) C_{2v} : shows a single loop with a minimum at $T=0$. (b) C_{2v} : shows a single loop with a minimum at $T=0$ and a small loop below it. (c) Σ : shows a figure-eight shape with a minimum at $T=0$. (d) C_{2v} : shows a single loop with a minimum at $T=0$ and a small loop below it.

Figure 2-1. The string running through this modified periodic table puts all the elements in sequential order according to the Mendeleev number. (Pettifor, 1988a). Note that group IIA elements Be and Mg have been grouped with IIB, divalent rare earths have been separated from trivalent, and Y has been slotted between Tb and Dy.

Intermetallic compounds: Valence compounds

I	II											III	IV	V	VI		
H																He	
Li	Be											B	C	N	O	F	Ne
Na	Mg											Al	Si	P	S	Cl	Ar
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn

metals from groups I, IV, V, VI

e.g.:



decreasing
electronegativity

Intermetallic phases: Hume-Rothery Phases

Hume-Rothery rule:

„electron phases“

compare rules for solid solution formation

similar structures for similar valence electron concentration
(VEC)

VEC = number of VE/atom of the phase, also (e/a) ratio

VEC = $3/2$ (21/14): „bcc“, β structures

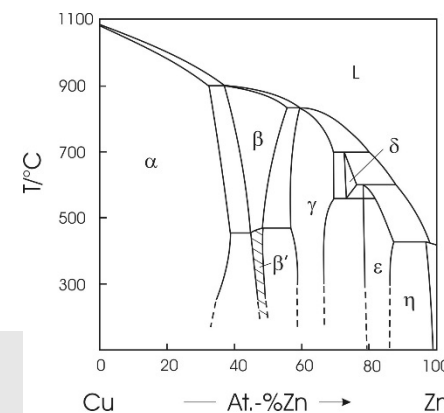
VEC = 21/13: γ structures

VEC = $7/4$ (21/12): „hex“ ϵ structures

e.g.: CuZn, CuBe, NiAl, ...

e.g.: Cu_5Zn_8 , Cu_9Al_4 , ...

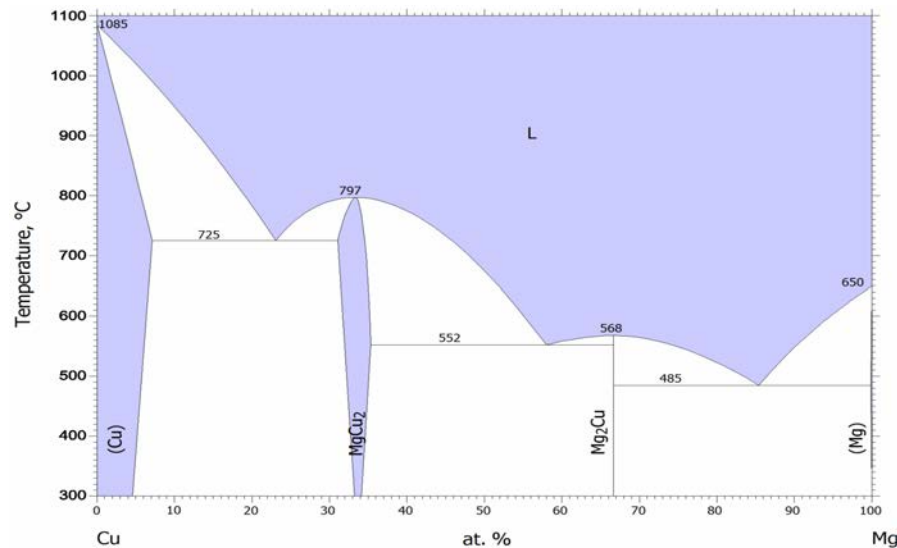
e.g.: CuZn_3 , CuSn_3 , ...



Laves phases (AB_2)

H																	He
Li	Be											B	C	N	O	F	Ne
Na	Mg											Al	Si	P	S	Cl	Ar
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn

frequently Laves-phases and related phases (e.g. $MgCu_2$, $NbFe_2$, $TiFe_2$, $TbFe_2$)



Laves-phases:

determined by ratio of atom radii

e.g. $1/1.225 \rightarrow AB_2$ - compounds

densely packed (e.g. 0.71) (bcc 0.682)

- usually hard, brittle
- undesired in superalloys
- interesting magnetic material ($TbFe_2$)

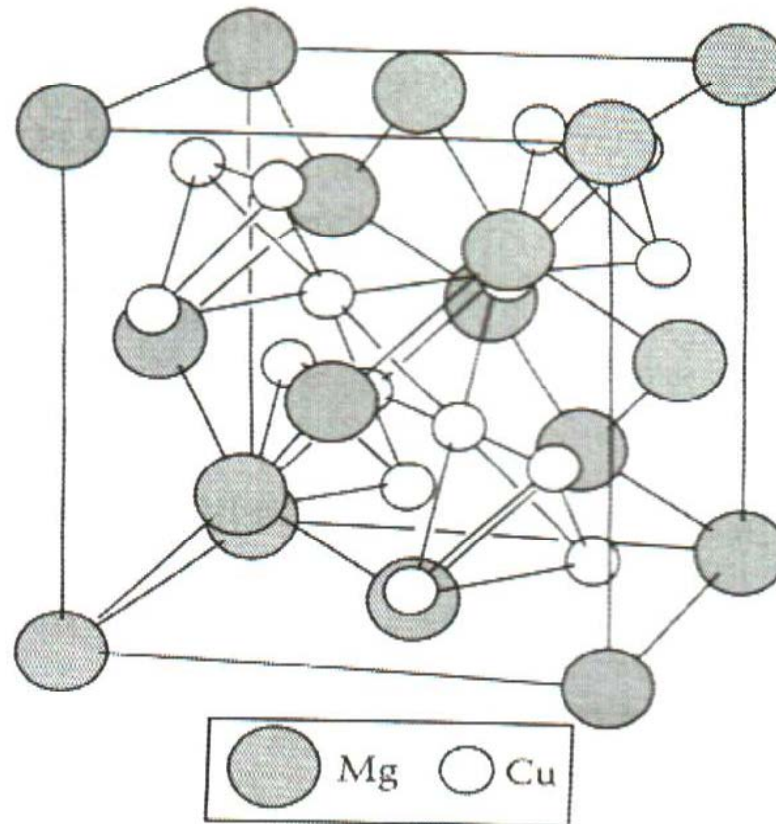
Intermetallic compounds: Laves-Phases

determined by the ratio of atom radii of components

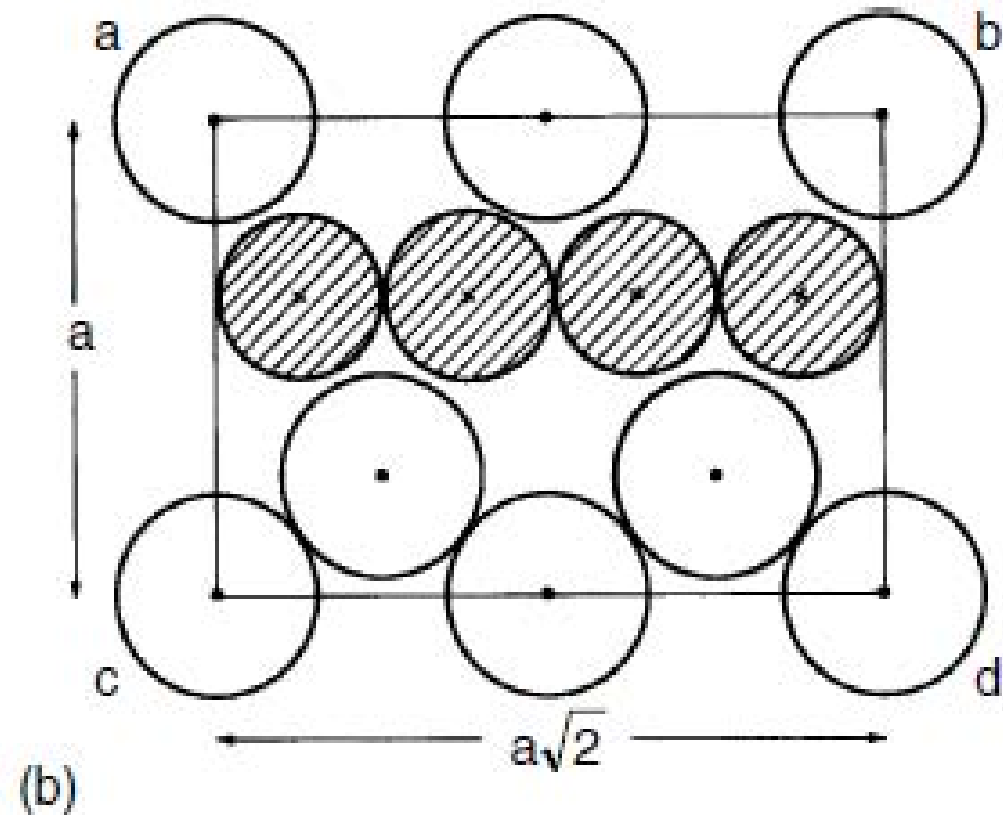
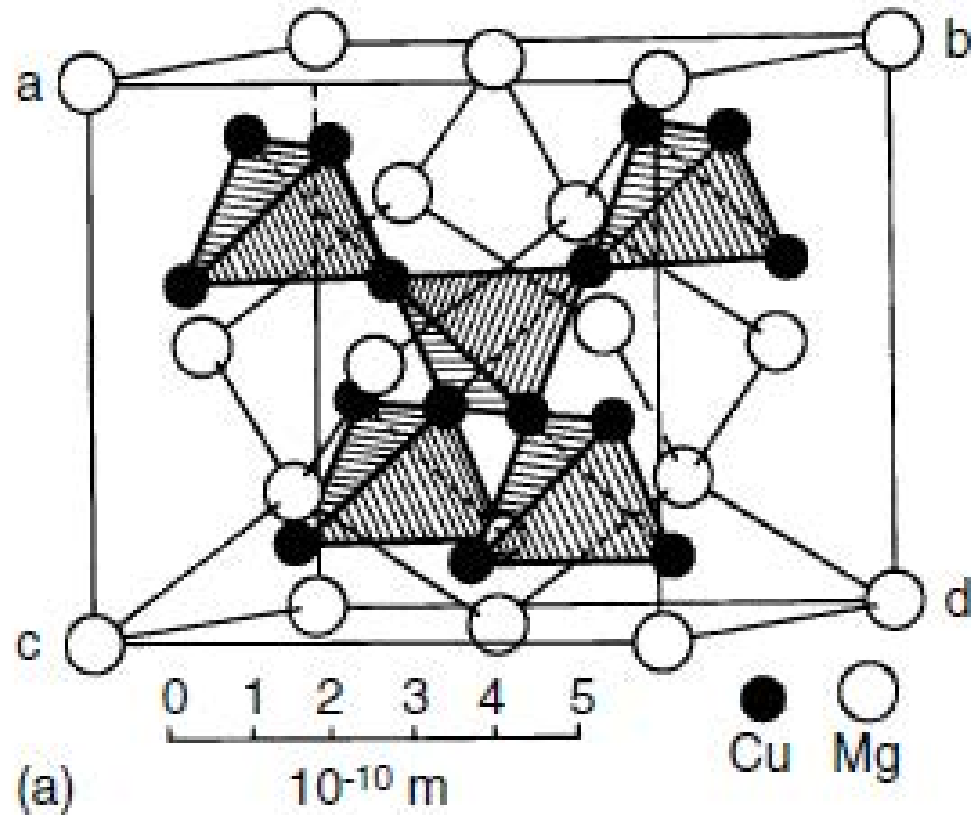
e.g. radii ratio 1/1.225 $\rightarrow$ AB₂-compounds

more densely packed (e.g. 0.71) than bcc (e.g. 0.682)

MgCu₂



Unit cell of the Laves phase MgCu_2



(110) plane

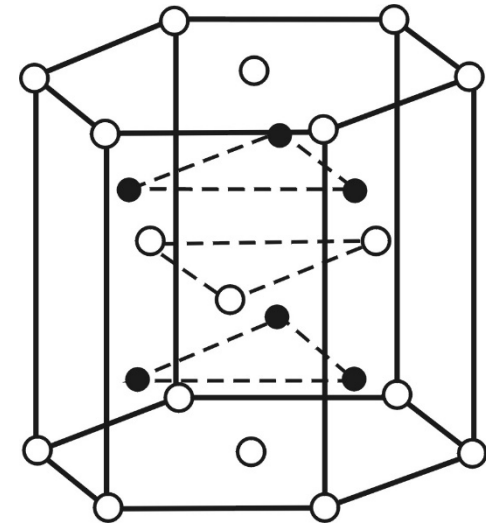
Intermetallic compounds: NiAs-compounds

H																	He
Li	Be											B	C	N	O	F	Ne
Na	Mg											Al	Si	P	S	Cl	Ar
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn

Compounds of: Se, Te, Sn, Sb, As, Bi
and transition metals: Fe, Cr, Co, Ni, Mn

NiAs:

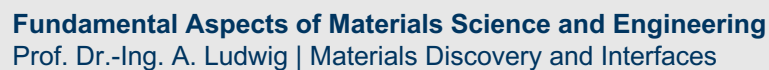
As-atoms form densest packed partial lattice
Ni occupies octahedral sites



○ As
 ● Ni

Fundamental Aspects of Materials Science and Engineering
Prof. Dr.-Ing. A. Ludwig | Materials Discovery and Interfaces

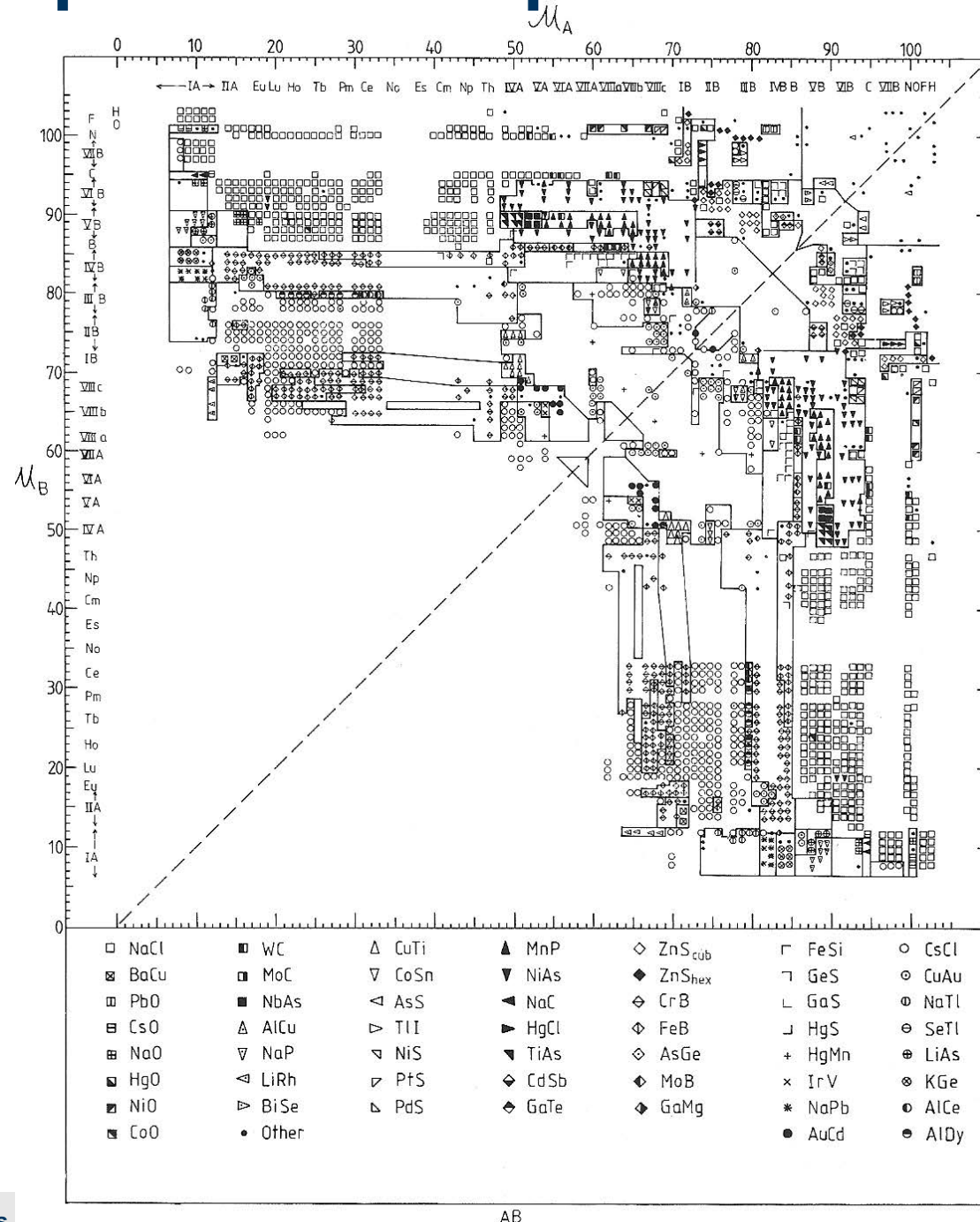
46



Source: Intermetallic Chemistry

Pettifor's map for AB compounds

AB structure map
(Pettifor 1988)



Pettifor's map for AB₂ and AB₃ compounds

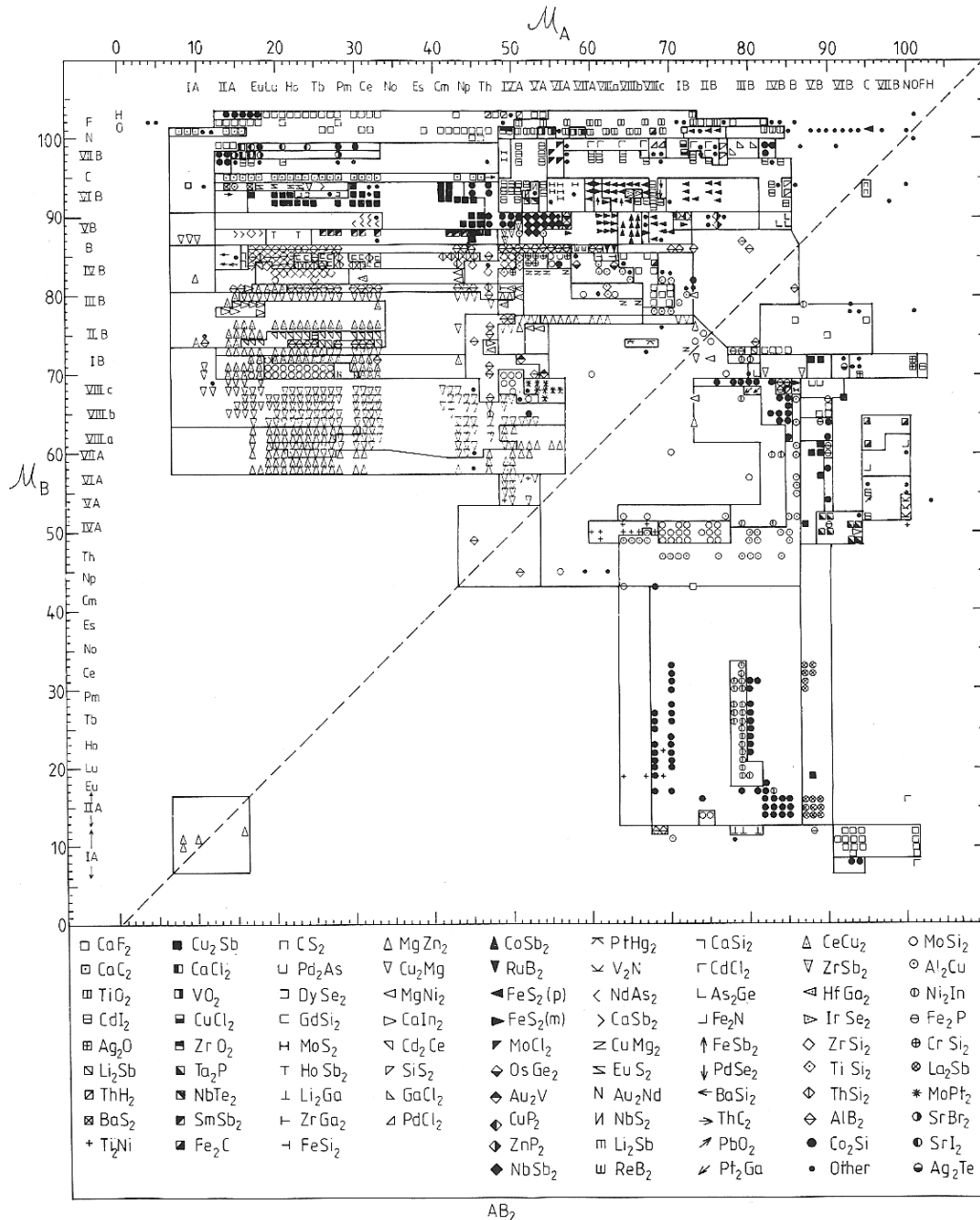


Figure 2-4. The AB₂ structure map (Pettifor, 1988 a).

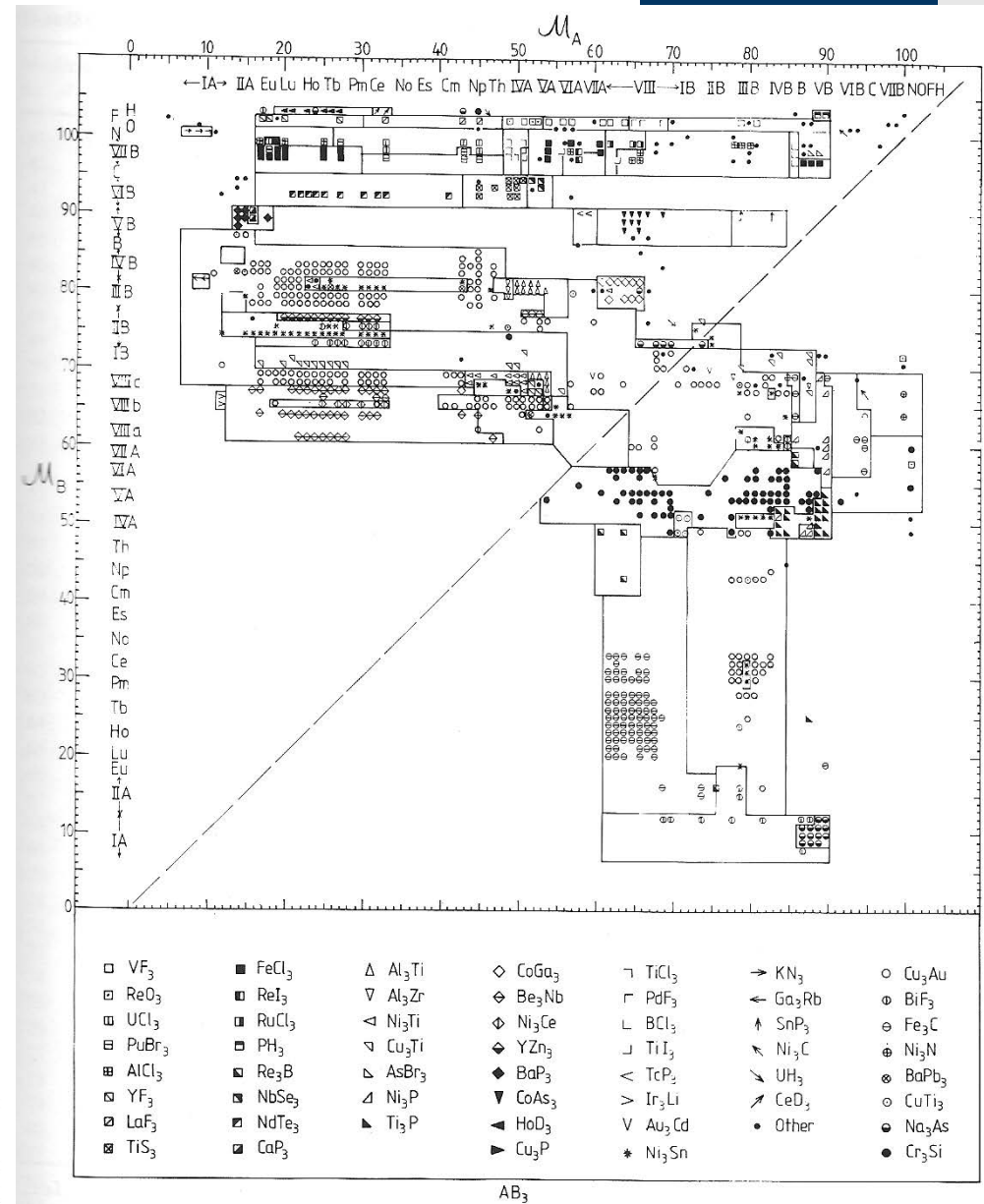
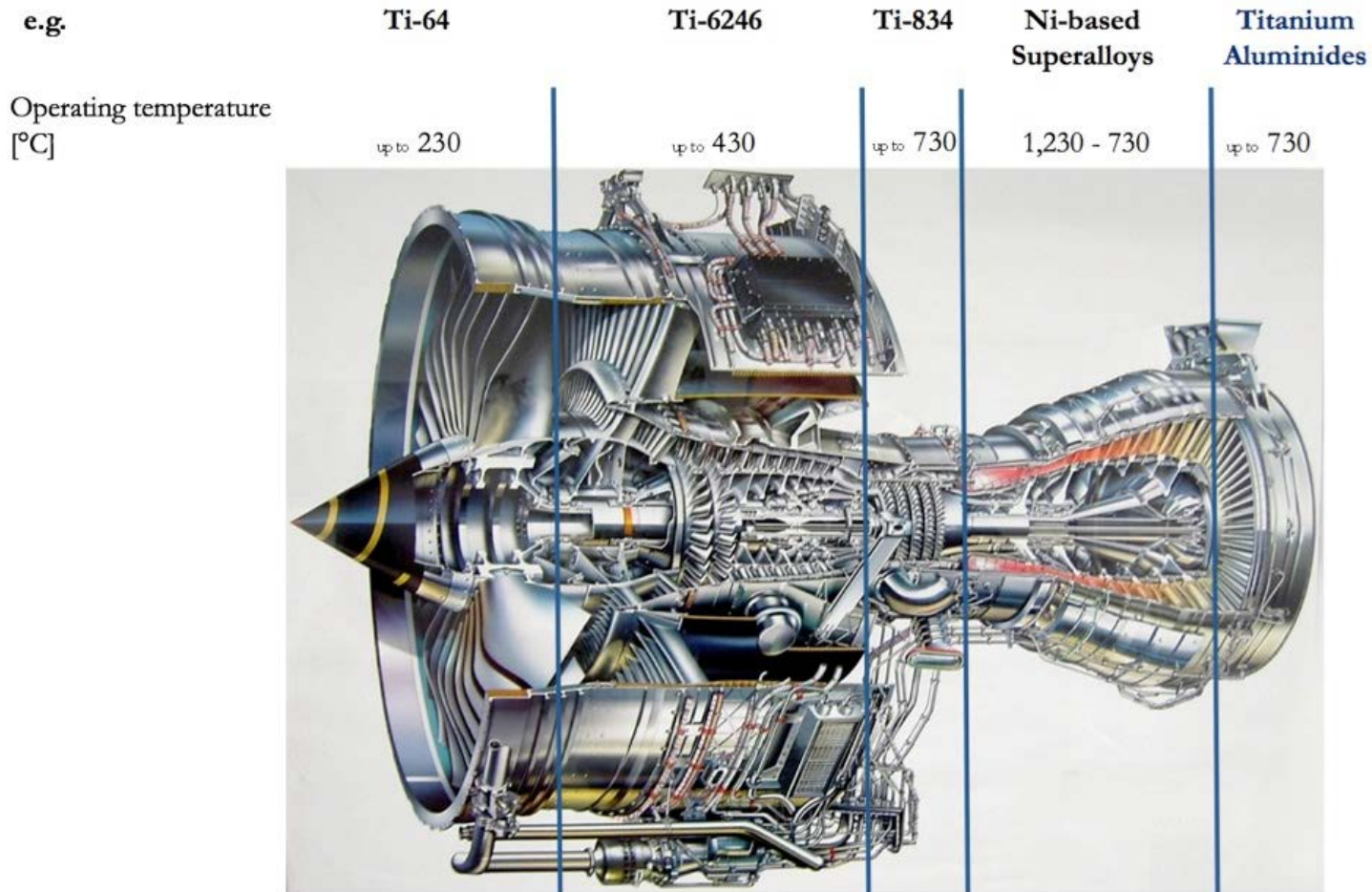
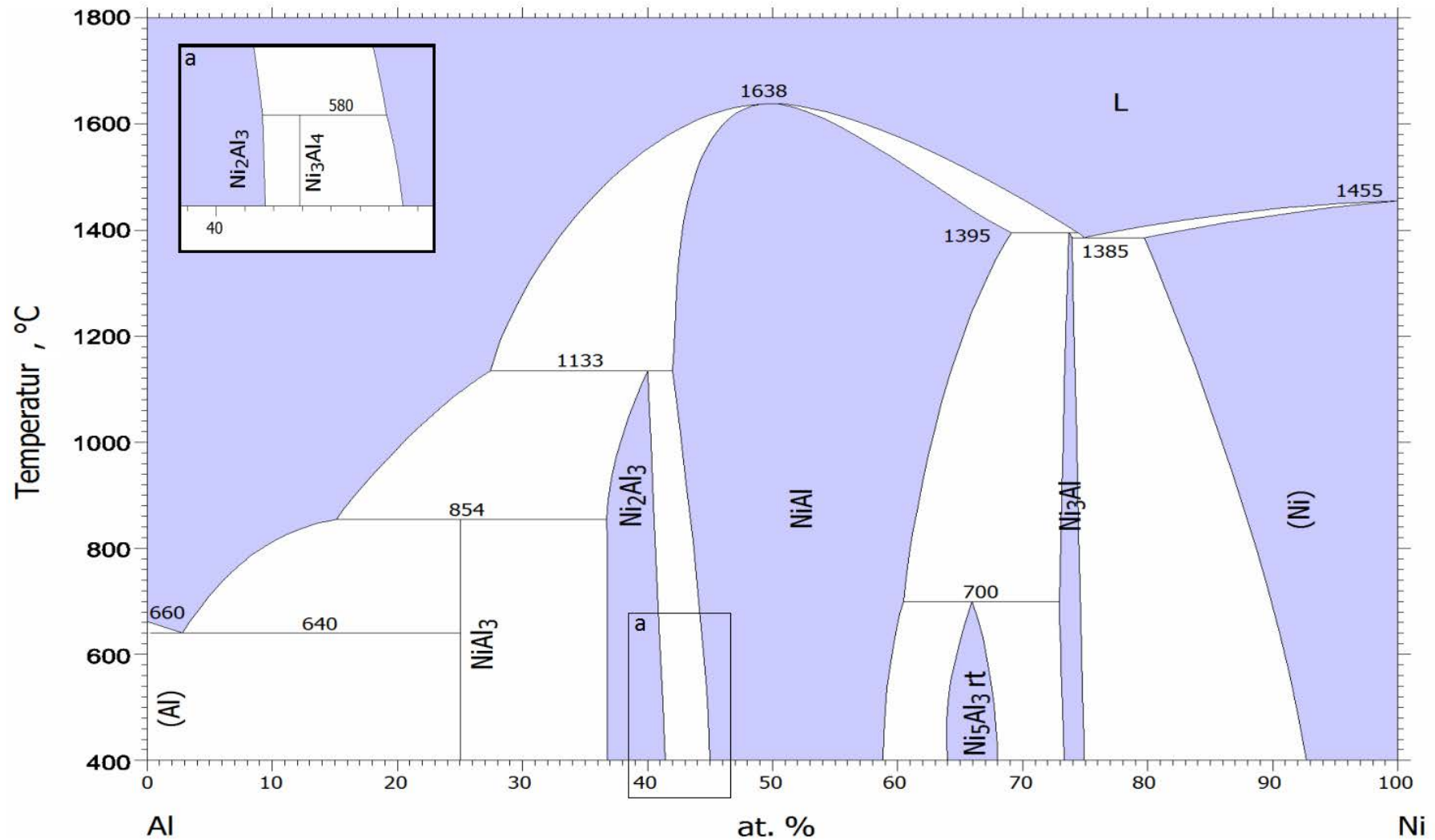


Figure 2-5. The AB₃ structure map (Pettifor, 1988 a).

Aluminides: Intermetallic phases for application in turbines

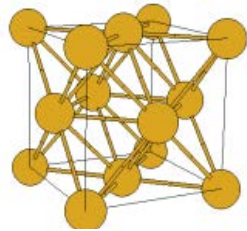


Intermetallic phases: Nickel aluminides

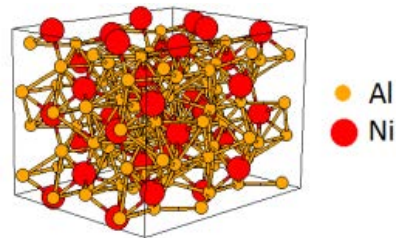


Intermetallic phases: Nickel aluminides

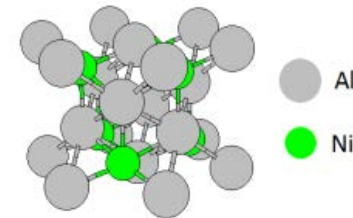
Crystal structures in the system Ni-Al



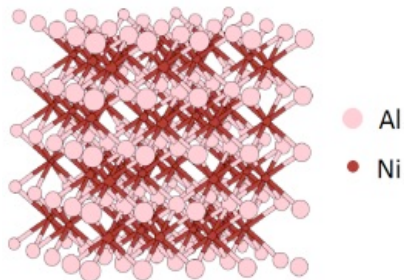
Ni/Al: face-centered cubic
Fm3m (225)



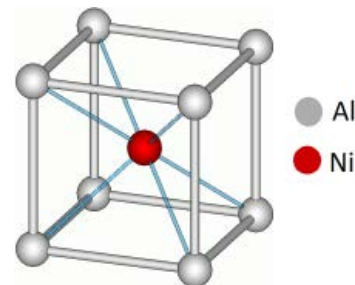
NiAl₃: orthorhombic Pnma (62)



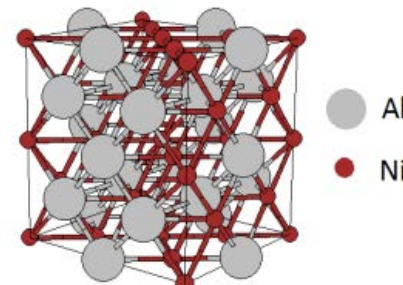
Ni₂Al₃: trigonal P3m1 (164)



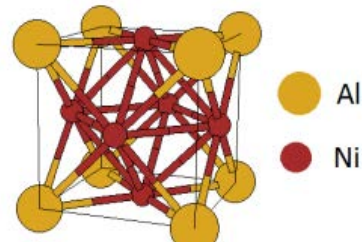
Ni₃Al₄: cubic Ia3d (230)



NiAl: body-centered cubic Pm-3m (221)



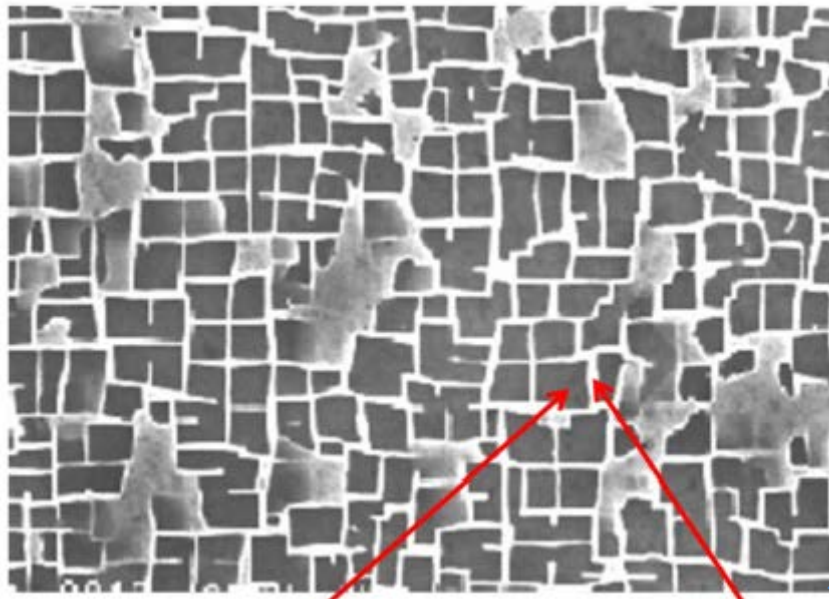
Ni₅Al₃: orthorhombic Cmmm
(65)



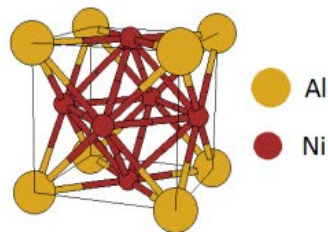
Ni₃Al: face-centered cubic
Pm-3m (221)

Intermetallic compounds: Nickel aluminides

application: Ni-based superalloys

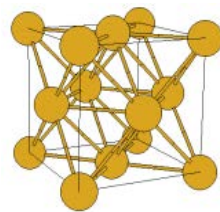


γ' -precipitate

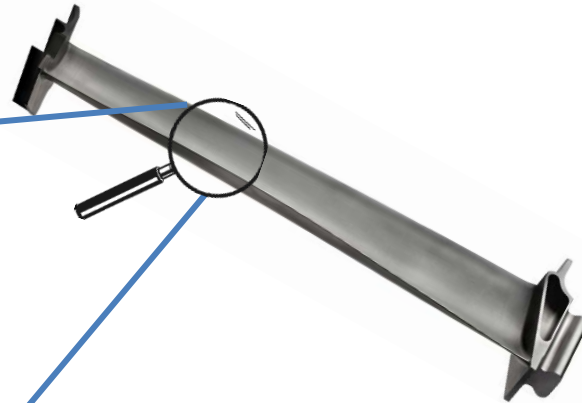


Ni_3Al : face-centered
cubic $\text{Pm}\bar{3}\text{m}$ (221)

γ -matrix

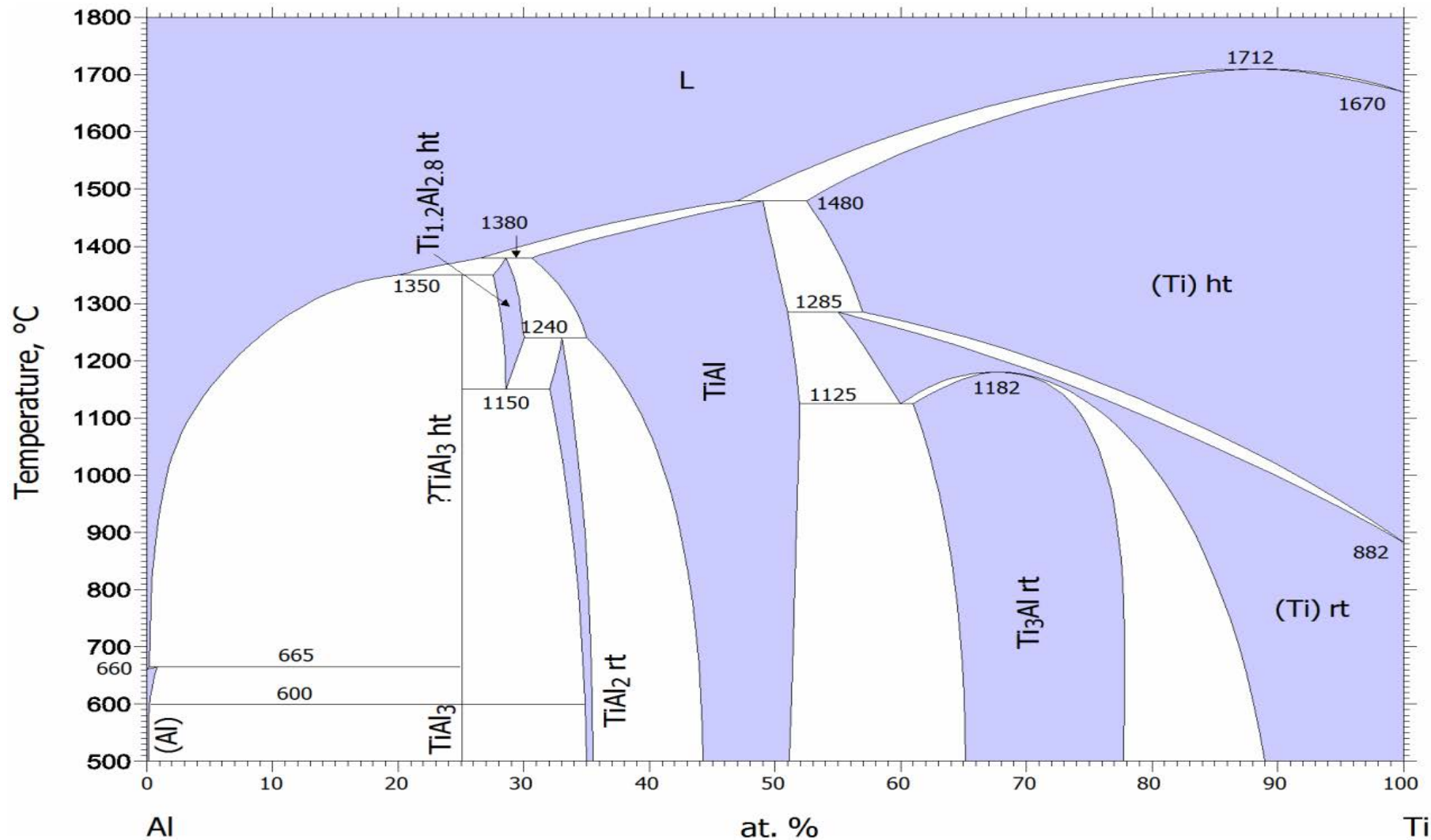


Ni/Al : face-centered
cubic $\text{Fm}\bar{3}\text{m}$ (225)



Intermetallic compounds: Titanium aluminides

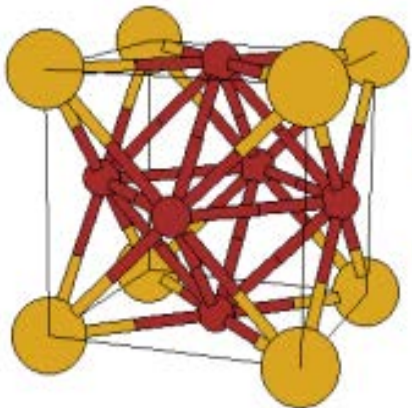
Phase diagram of the system Ti-Al



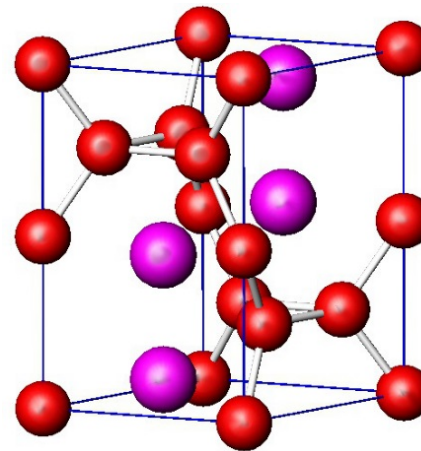
Intermetallic phases: Titanium aluminides

Phases in the system Ti-Al

- Most important for applications: TiAl and Ti₃Al
- other phases are not used in commercial alloys

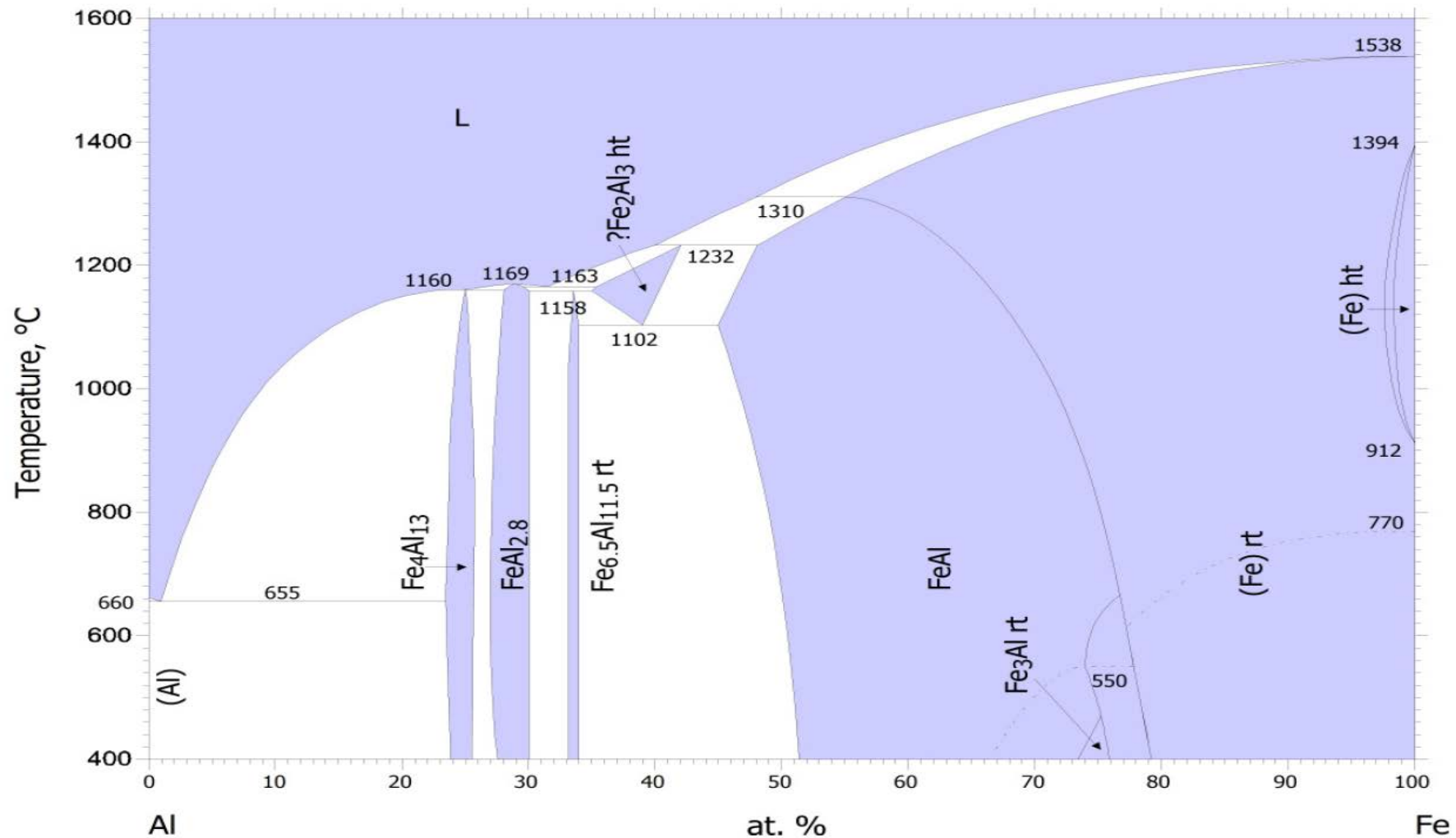


γ - TiAl (tetragonal)

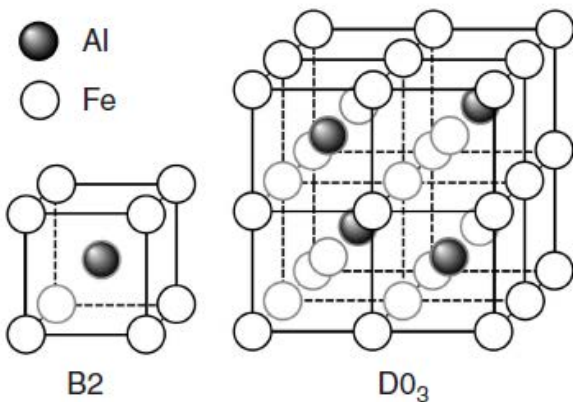
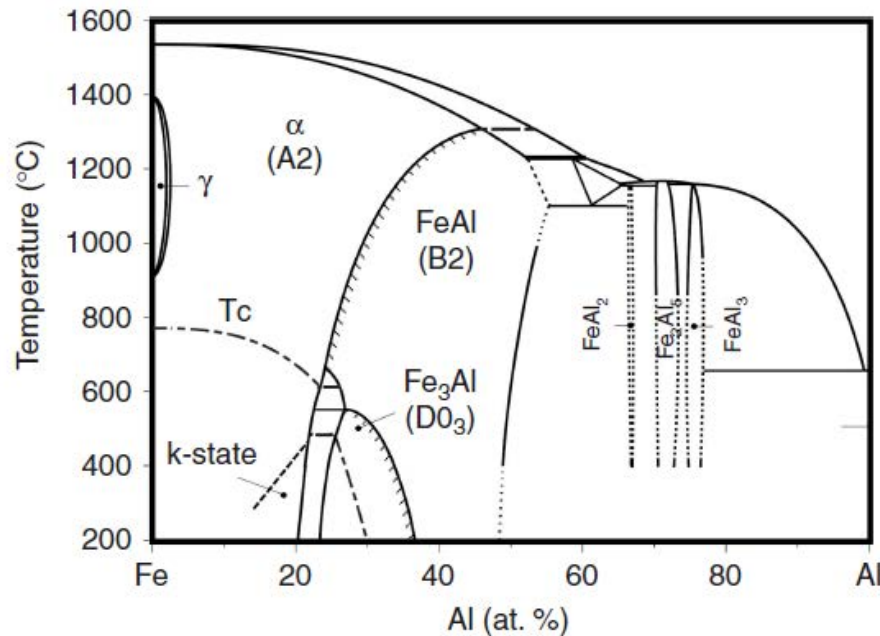


α_2 Ti₃Al (cubic)

Intermetallic phases: Iron aluminides



Intermetallic phases: Iron aluminides



Phases in Fe-Al

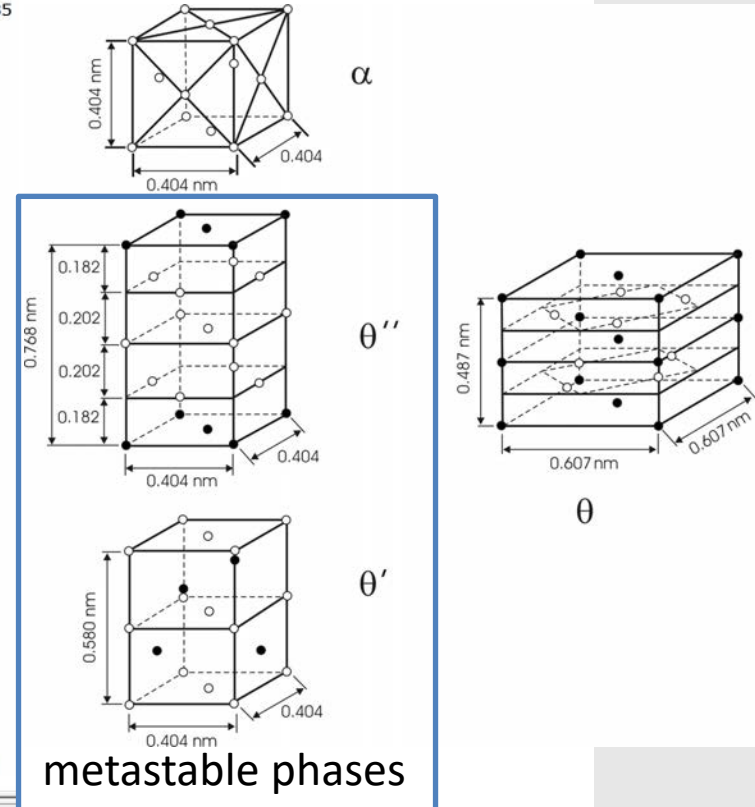
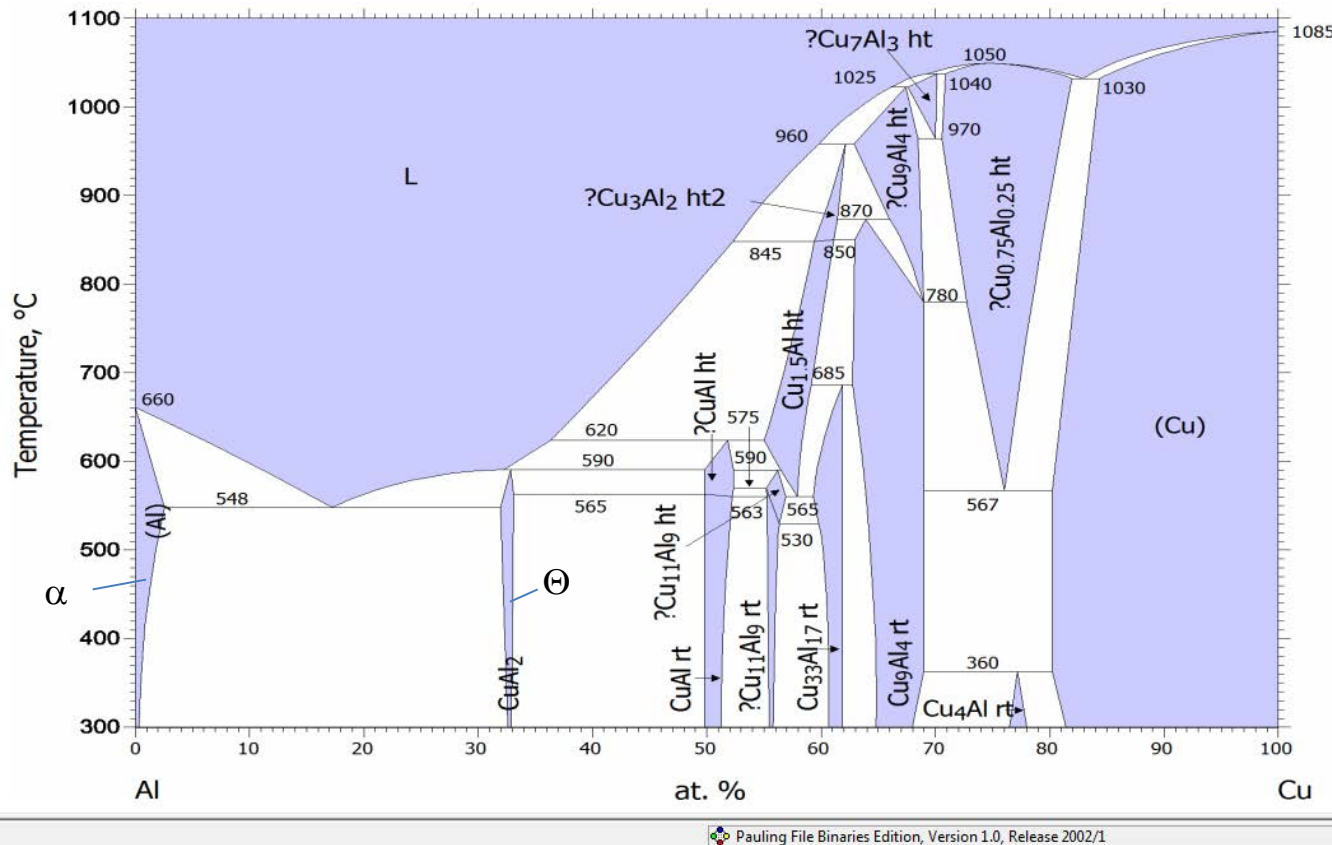
- most important: Fe_3Al and $FeAl$
- high electrical resistance and good corrosion resistance
- applications: e.g. heating elements
- problem: brittleness



3 Fe-Al binary phase diagram and crystal structures of B2 and D0₃ [24].

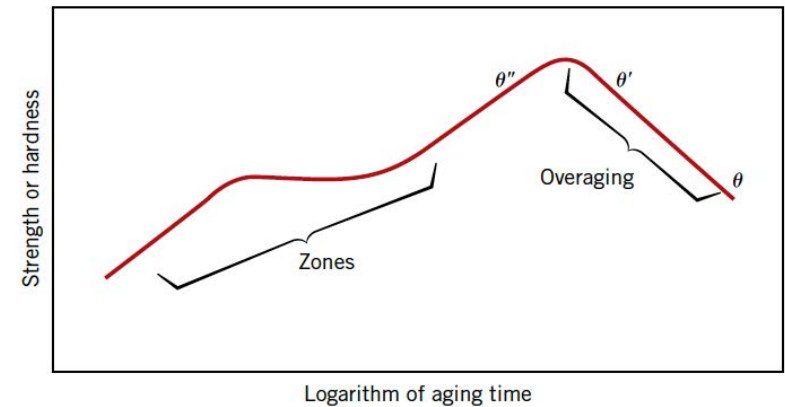
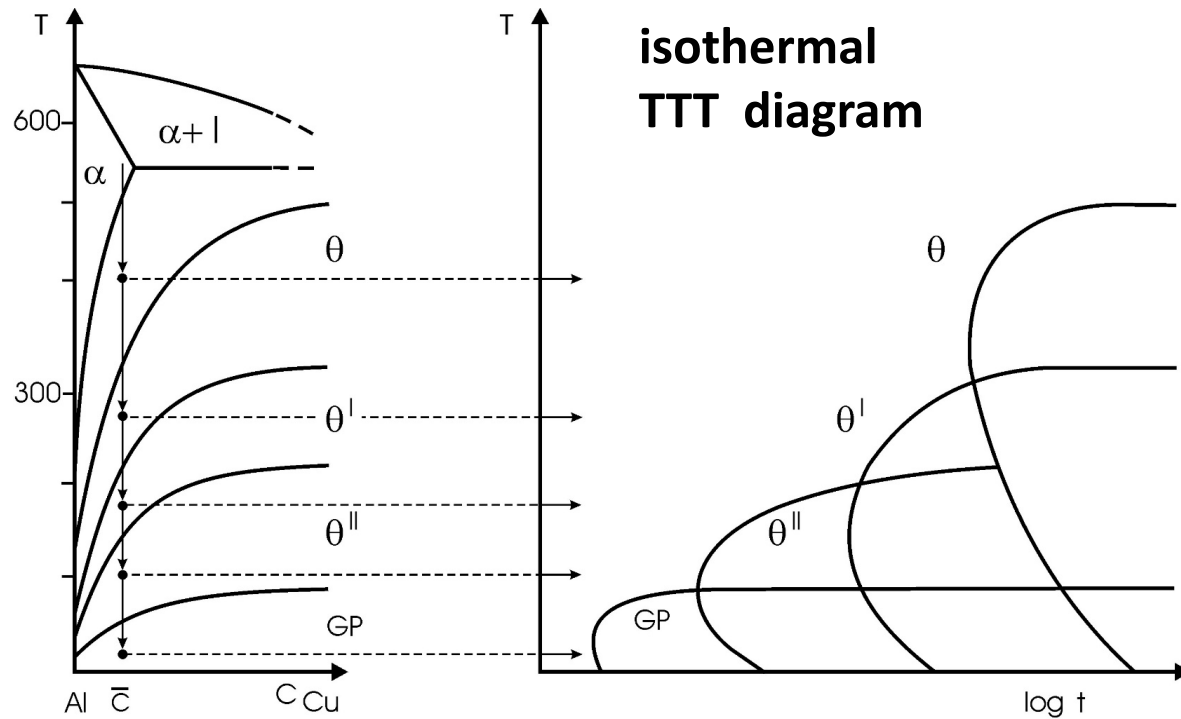
[S.C. Deevi, V.K. Sikka: Nickel and iron aluminides: an overview on properties, processing, and applications]

From metastability to stability: Precipitation strengthening in the system Al-Cu by age hardening



Important for lightweight constructions, e.g. aerospace,
high strength Al-alloys

From metastability to stability: Precipitation strengthening in the system Al-Cu

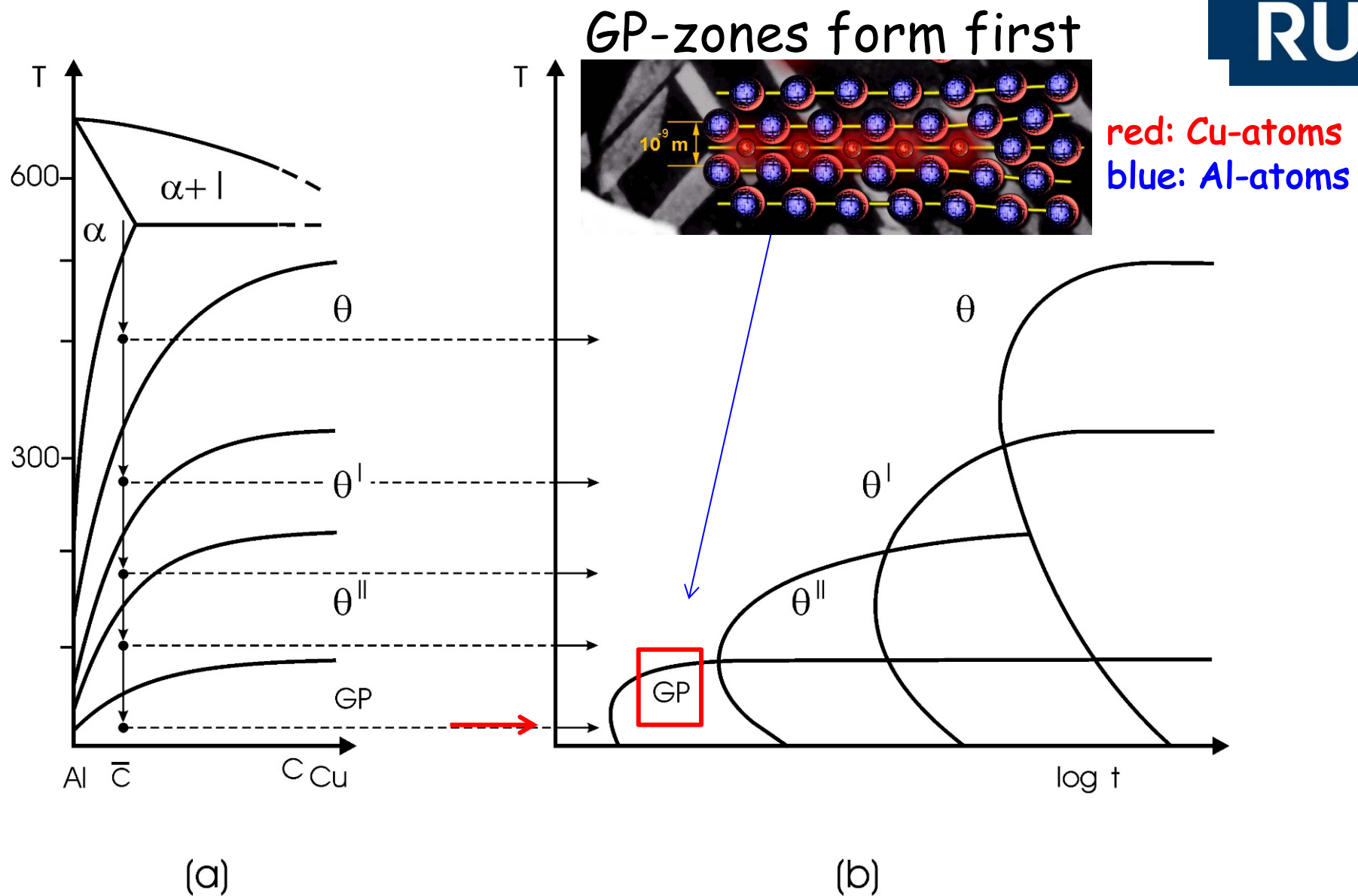


(a)
Phase diagram with cooling/holding steps

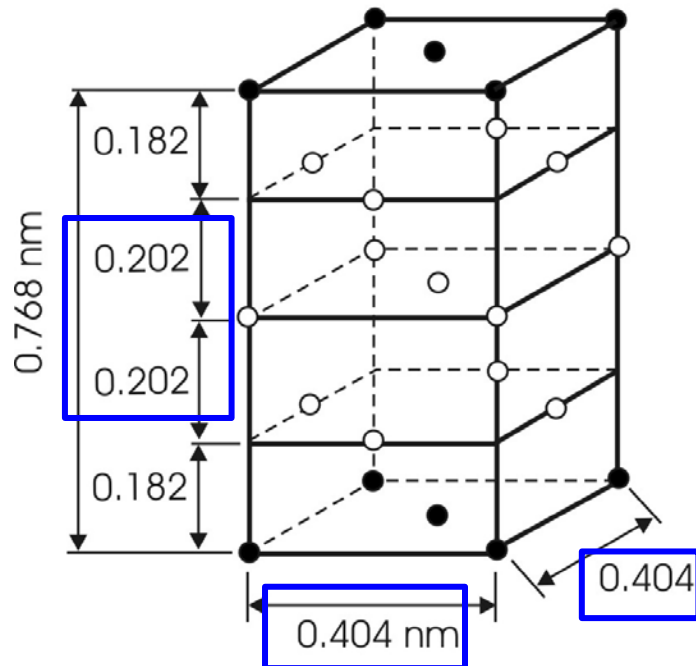
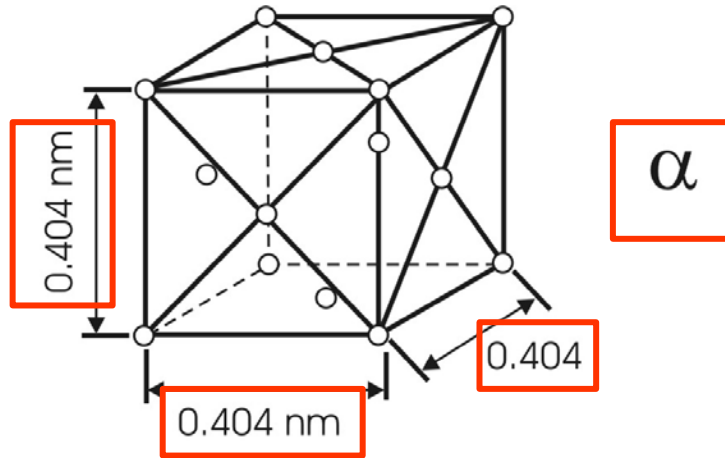
(b)

what fits best
(from a point of view of the
lattices of the parent and the product phase)
forms first
GP: Guinier-Preston zones

Early stage of nucleation



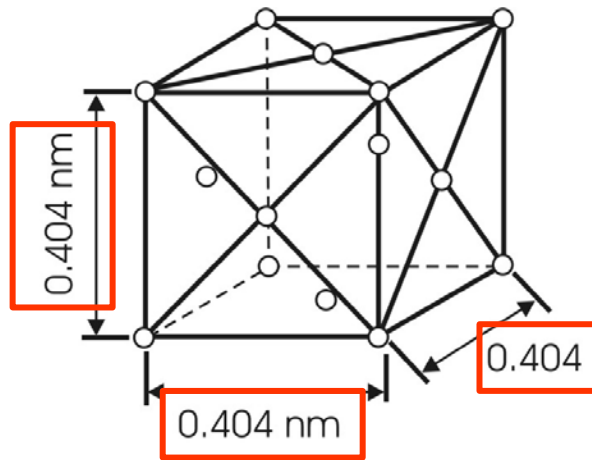
Al and θ''



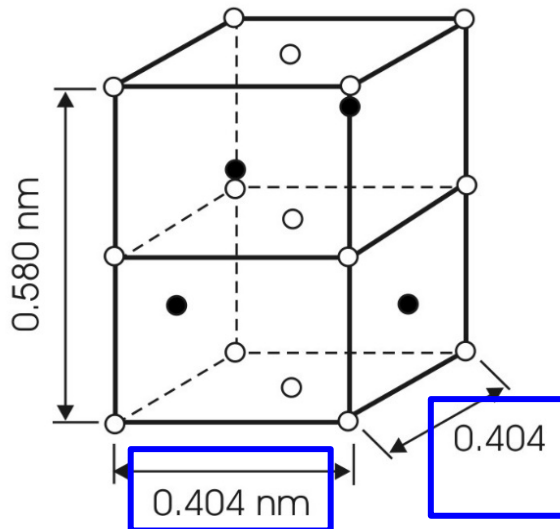
θ''

lattice fits very well
into **Al-lattice**

Al and θ'



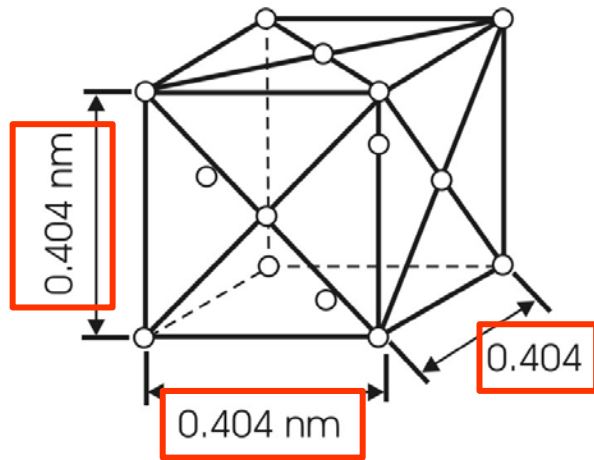
α



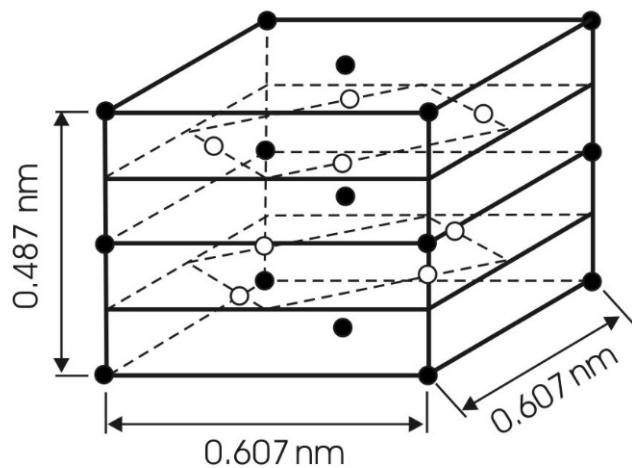
θ'

θ' still fits rather well

Al and θ



α



θ

crystal lattice of
equilibrium phase θ
does not fit well

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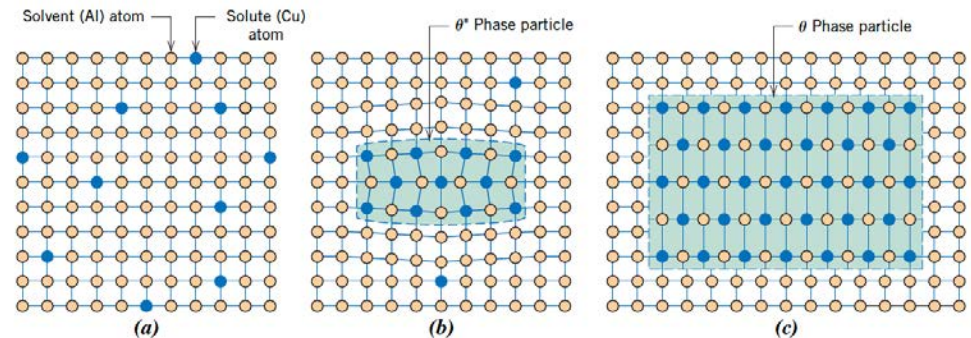
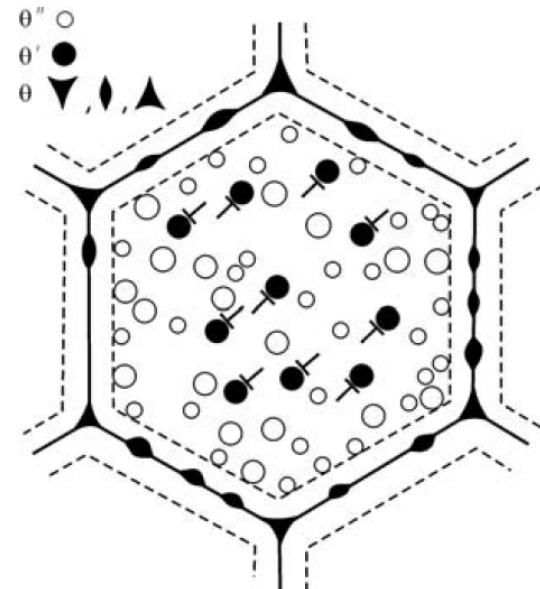


Figure 11.26 Schematic depiction of several stages in the formation of the equilibrium precipitate (θ) phase. (a) A supersaturated α solid solution. (b) A transition, θ' , precipitate phase. (c) The equilibrium θ phase, within the α -matrix phase.



Heusler phases

- discovered by Heusler in 1903
- ferromagnetism in a ternary phase of non-ferromagnetic components

Heusler discovered that an alloy of the non-ferromagnetic elements Cu, Mn and Al shows ferromagnetism (Cu_2MnAl)

Über magnetische Manganlegierungen; von Fr. Heusler.

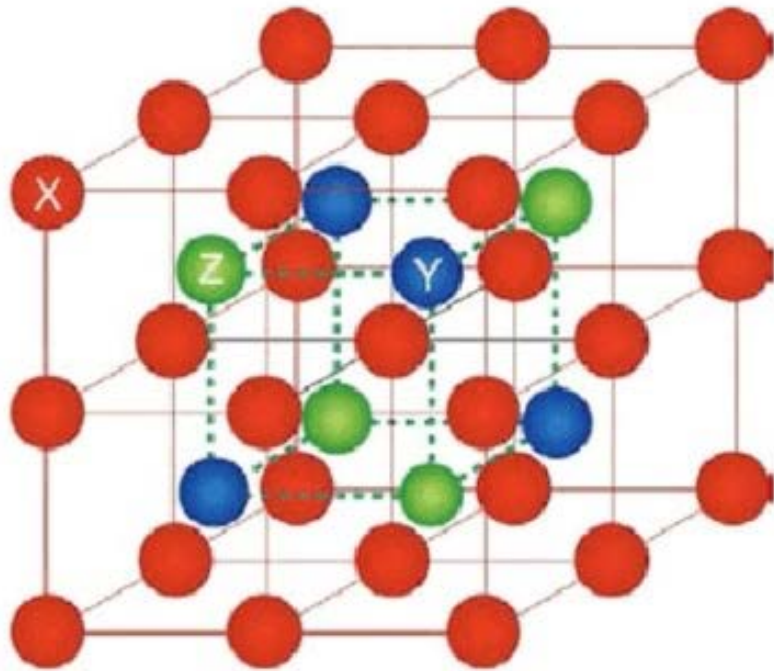
(Niedergelegt im Archiv der Deutschen Physikalischen Gesellschaft am
18. Juni 1901, der Öffentlichkeit übergeben in der Sitzung vom 12. Juni 1903.)

(Vgl. oben S. 217.)

Ich habe gefunden, daß im Gegensatz zu den unmagnetischen Eigenschaften des Manganmetalles sowie des Mangankupfers gewisse andere Legierungen des Mangans stark magnetisierbar sind und diese Eigenschaft auch behalten, wenn man den Legierungen Kupfer und andere an sich unmagnetische Metalle zusetzt. Die

Heusler phases

$L2_1$ -type X_2YZ



X, Y transition metals
Z element of main groups

4 overlapping fcc lattices

X: (0,0,0) und ($\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$)

Y: ($\frac{1}{4}$, $\frac{1}{4}$, $\frac{1}{4}$)

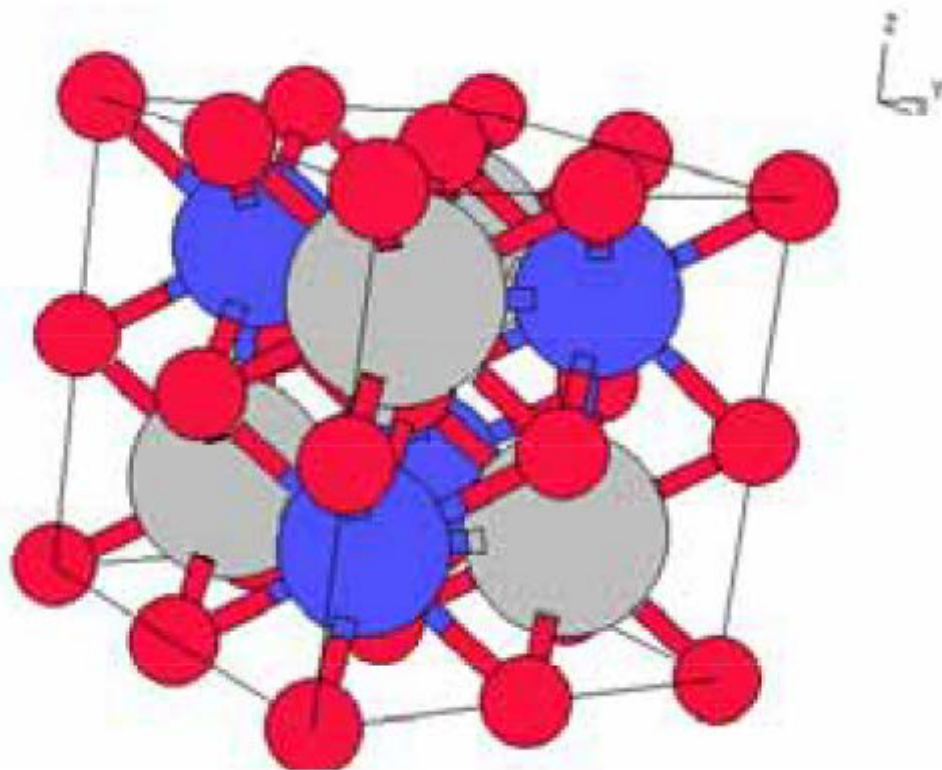
Z: ($\frac{3}{4}$, $\frac{3}{4}$, $\frac{3}{4}$)

full Heusler structure

(„combine two different B2 cells“)

The ferromagnetic shape memory alloy Ni_2MnGa

Ni_2MnGa : Heusler phase



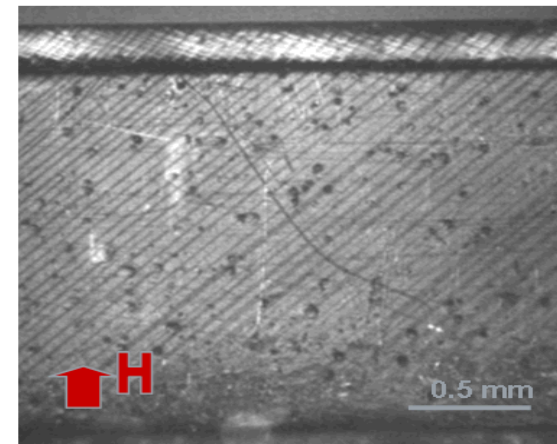
$L2_1$ (austenite)

martensitic transformation:

„fcc“ austenite to tetragonal martensite



MSM  www.MagneticShape.de



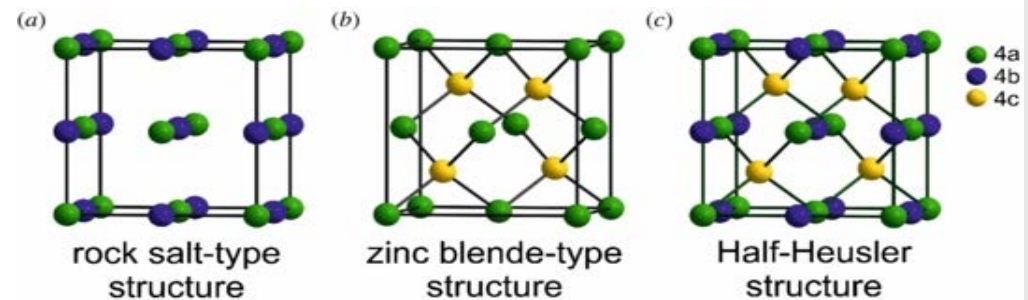
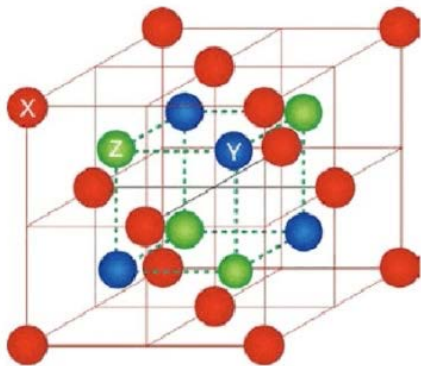
© Ryan Lai & Jeffrey McCord, IFW Dresden

Source: www.magneticshape.de/

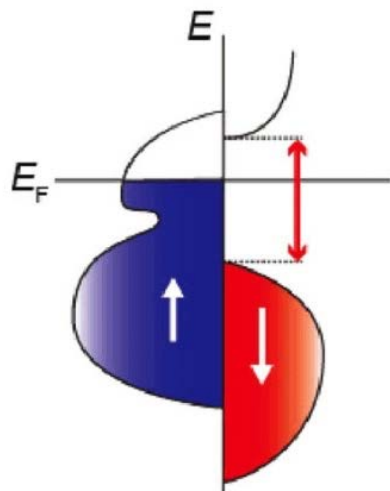
Half-Heusler phases

Half-Heusler-structure = Full-Heusler-structure with unoccupied sub-lattice

$C1_b$ -type XYZ



Casper et al., Semicond. Sci. Technol. (2012)



band gap

semiconducting if VEC (valence electron count) is 18

thermoelectric half-Heusler phase NiTiSn

Source: Wen et al., Nature (2015)

Isothermal sections of ternary phase diagrams

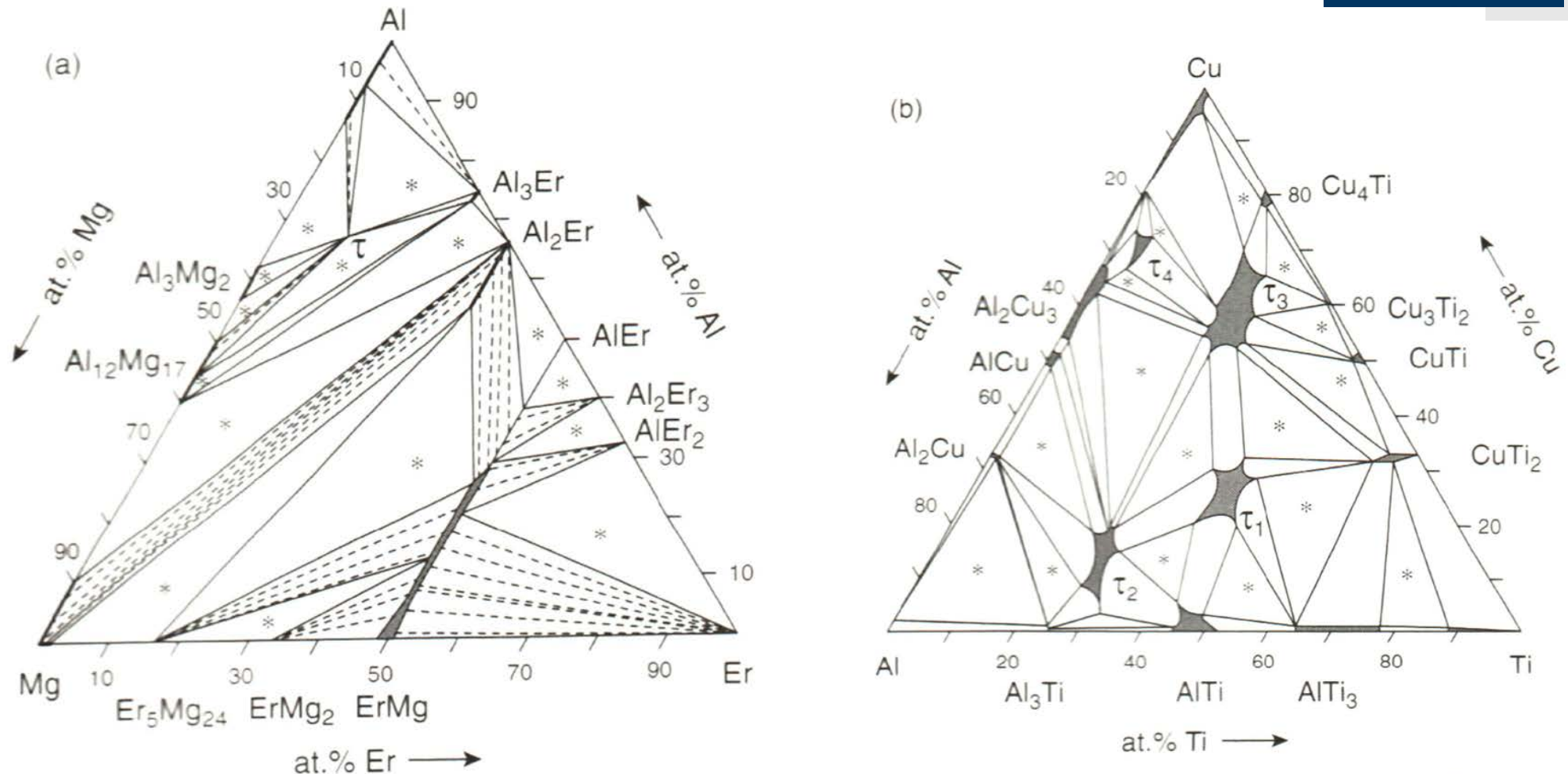


Figure 2.29. Isothermal sections of ternary phase diagrams: (a) Al-Er-Mg system at 400°C, Saccone *et al.* (2002) and, (b) Al-Cu-Ti system at 540°C from Villars *et al.* (1995). A number of single-phase regions (dark grey) may be noticed, both extending from binary compounds and as ternary intermediate phases (τ) in the Al-Er-Mg system and the four phases τ_1 , τ_2 , τ_3 and τ_4 in the Al-Cu-Ti system. The three-phase fields are marked by an asterisk, in the Al-Er-Mg system a few tie-lines are indicated in the two-phase fields.

Ternary intermetallic phases

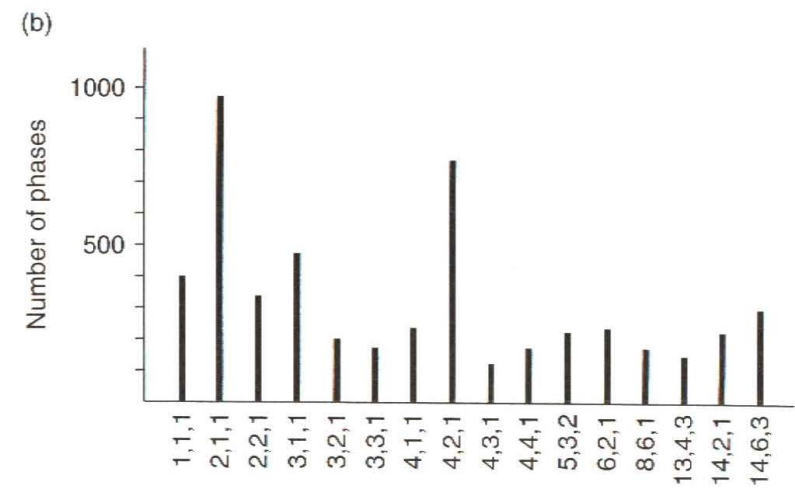
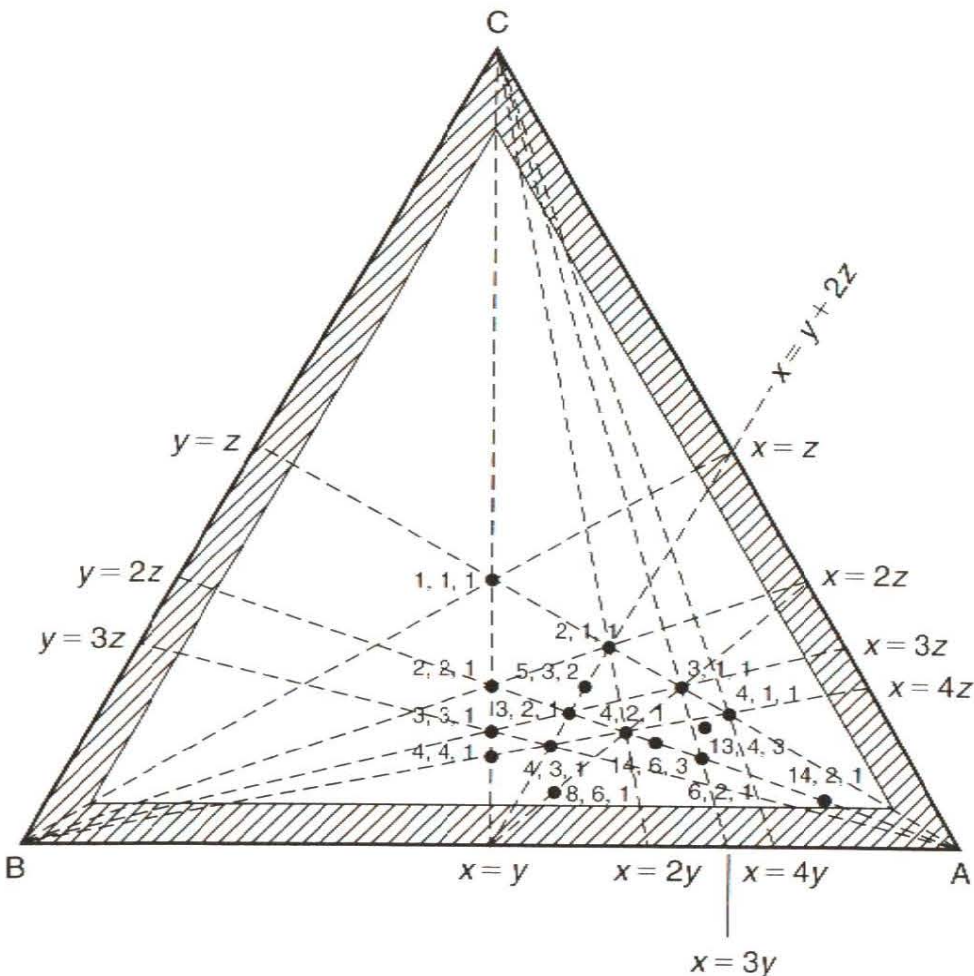


Figure 7.2. Distribution of the more common ternary intermetallic phases according to their stoichiometry. (a) Shows the position of $A_xB_yC_z$ phases in a representative portion of a general composition triangle. The hatched region corresponds to composition values for which in general ternary compounds are very seldom observed. (b) An indication is given about the number of phases reported by Rodgers and Villars (1993) for different typical stoichiometries.

Hydrides, carbides, borides, nitrides

Metal hydrides:

e.g. Mg-H, TiH₂

Metal borides:

e.g. MgB₂

Metal nitrides:

e.g. TiN

Metal carbides

e.g. TiC, WC, Fe₃C

Interstitial compounds:

transition metals + H,C,B,N

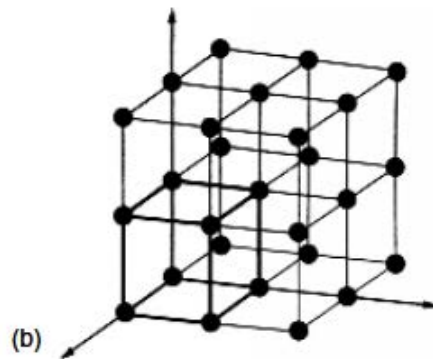
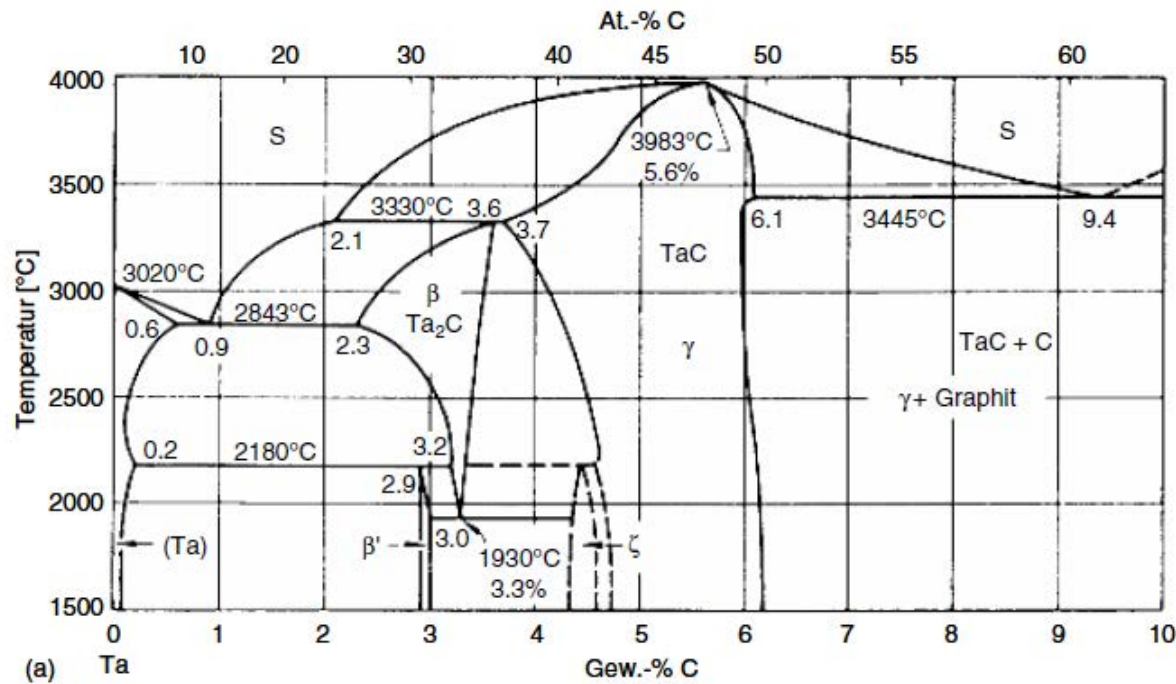
very hard, very high T_m

$C > C_{\text{solv}}$:

small atoms are still in interstitial positions

but overall crystal structure changes

Carbides



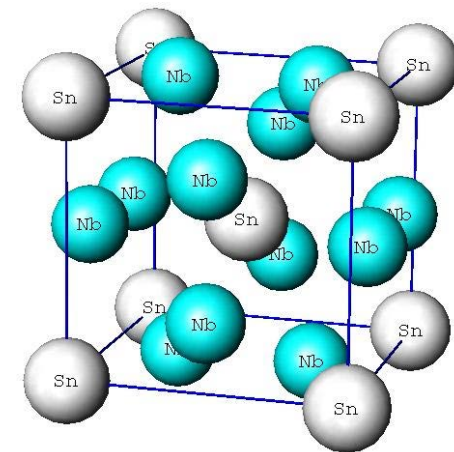
Hägg-
phases

here: TaC
(fcc)
 Ta_2C (h)

Abbildung 4.47. (a) Zustandsdiagramm des Systems Ta-C mit den Hägg-Phasen TaC (kfz) und Ta_2C (hexagonal). (b) Gitter des TaC; die C-Atome sitzen auf allen oktaedrischen Zwischengitterplätzen im kfz Gitter (NaCl-Struktur) [4.1].

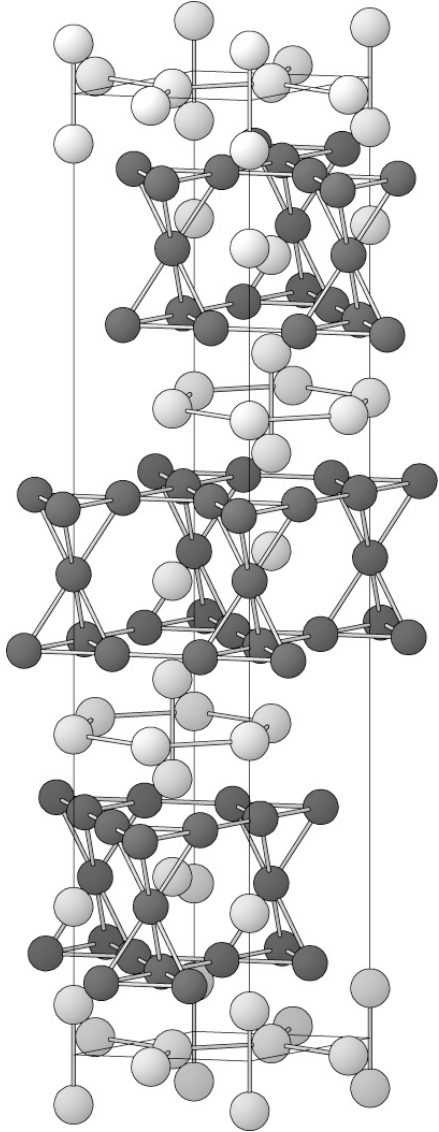
Topologically close packed (TCP) phases: A15

- complex cubic structure derived from A2 with 8 atoms in unit cell
- Typical example for TCP, which result from stacking of polyhedra of various shapes to accomodate atoms of different sizes
- Laves phases (C14, C15, C36) can be derived from A15 by crystallographic operations
- Important superconductors: Nb_3Sn , Nb_3Ge
- Problem: brittleness

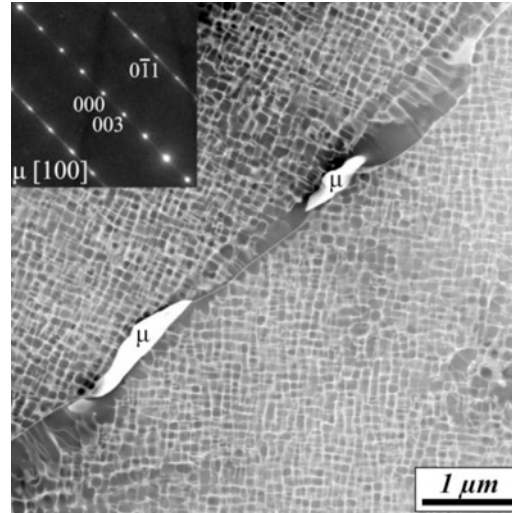


- TCP structures allow closer stacking of atoms with different sizes than geometrically close packed structures (A1, A3)
- intermetallic phases with TCP structure: Frank-Kaspar phases

Topologically close packed (TCP) phases: μ -phase

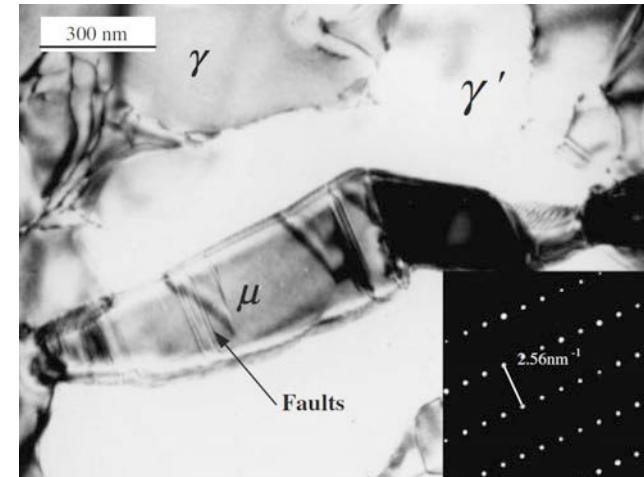


μ -phase in CMSX-4 Ni-base superalloy produced by selective electron beam melting



Alireza B. Parsa, Markus Ramsperger, Aleksander Kostka, Christoph Somsen, Carolin Körner and Gunther Eggeler: *Transmission Electron Microscopy of a CMSX-4 Ni-Base Superalloy Produced by Selective Electron Beam Melting*, *Metals* 2016, 6(11), 258

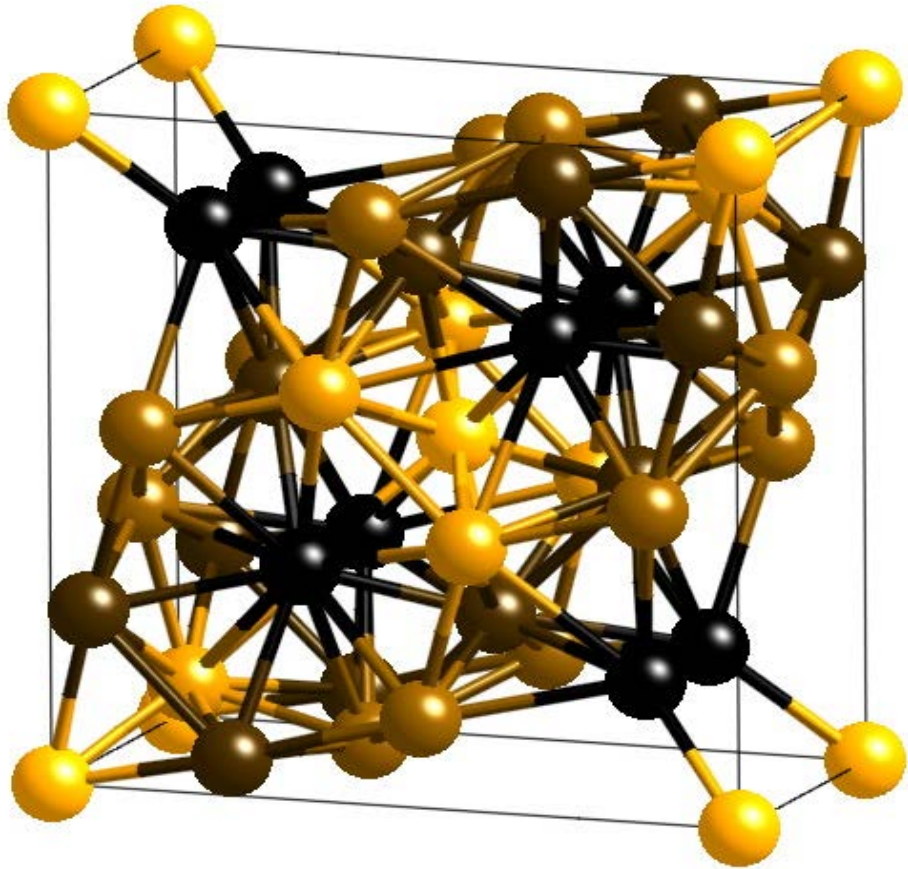
μ -phase in a Ni-base directionally solidified alloy



K. Zhao, Y. H. Ma, L. H. Lou and Z. Q. Hu: *Phase in a Nickel Base Directionally Solidified Alloy*, *Materials Transactions*, Vol. 46, No. 1 (2005) pp. 54 to 58

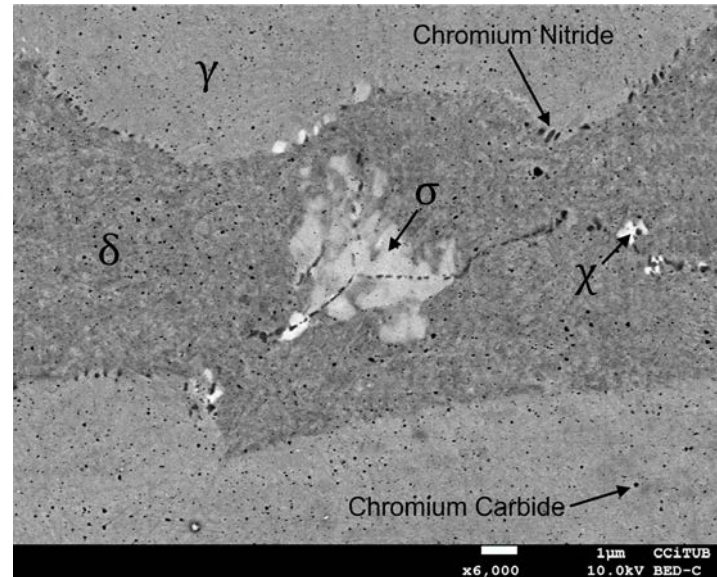
- hexagonal system
- typically forms in superalloys from Co and W $\rightarrow$ W_6Co_7

Topologically close packed (TCP) phases: σ -phase



<http://som.web.cmu.edu>

σ -phase phase in duplex stainless steels



Núria Llorca-Isern, Héctor López-Luque, Isabel López-Jiménez, Maria Victoria Biezma:
Identification of sigma and chi phases in duplex stainless steels, Materials Characterization
 Volume 112, February 2016, Pages 20-29

- tetragonal system
- typically forms in stainless steel and superalloys from Fe and Cr $\rightarrow$ FeCr

Questions

- (1) Define the terms solid solution, alloy and compound. Discuss the stoichiometry and the compositional existence range.
- (2) What is allotropy?
- (3) What do the abbreviations rt, ht and lt mean?
- (4) What is the difference between alloying and doping?
- (5) Explain the difference between an interstitial and a substitutional element.
- (6) Why are intermetallic compounds frequently brittle?
- (7) Name the intermetallic compound with the highest technical relevance in the Ni-Al system. What is its crystal structure?
- (8) What is a Heusler phase? Sketch a unit cell.
- (9) Cu_2MnAl was the first Heusler phase to be discovered. What special property does it have?
- (10) What is the structural difference between a half Heusler and a (full) Heusler phase?
- (11) What is a Pettifor map?
- (12) Name two important intermetallic compounds in the Fe-Al materials system. Where and why are they applied?
- (13) What does the abbreviation TCP mean?
- (14) What is the purple plague?
- (15) Name a Laves phase with magnetostrictive behavior.
- (16) Name an intermetallic compound in the Ni-Ti system frequently used in application. What property makes this compound valuable?
- (17) What is a Guinier-Preston zone?
- (18) What is an interstitial compound? Give two examples.

Literatur

